OER Electrocatalysts

Subjects: Electrochemistry Contributor: Shuai Hou

The oxygen evolution reaction (OER) is the efficiency-determining half-reaction process of high-demand, electricity-driven water splitting due to its sluggish four-electron transfer reaction. Tremendous effects on developing OER catalysts with high activity and strong acid-tolerance at high oxidation potentials have been made for proton-conducting polymer electrolyte membrane water electrolysis (PEMWE), which is one of the most promising future hydrogen-fuel-generating technologies. Electrochemical water splitting involves two heterogeneous multi-step half-reactions, which are referred to as the cathodic hydrogen evolution reaction (HER) and the anodic oxygen evolution reaction (OER). An important frontier in OER electrocatalysis research is the development of the rational design of catalysts. Most of the excellent OER catalysts with high activity and durability are not stable in acidic solutions. They are easily oxidized and decomposed in a strong acid system, which is one of the indispensable working conditions for PEMWE. Outstanding OER electrocatalysts should have excellent intrinsic activity and sufficient active sites, and these requirements are generally combined with simplicity and controllability.

Keywords: water electrolysis; acidic oxygen evolution reaction; electrocatalyst; OER activity

1. Background

Searching for sustainable, clean, and highly efficient energy is the main method for solving the energy crisis and environmental pollution problems brought about by the first and second industrial revolutions, which have built a modern and prosperous society based on carbon-based fuels [1][2][3][4]. Human actions have led to the carbon dioxide content in the atmosphere rising rapidly and exceeding 400 ppm currently, mostly originating from the burning of coal, oil, and gas [2]. The application of wind and solar power and other types of renewable electricity generation technologies seems to be an efficient way to fulfill the requirement of an energy revolution [9][7][8]. However, they are strongly intermittent in nature due to diurnal or seasonal variations [9]. Converting their energy to a zero-emission chemical energy carrier such as hydrogen is an alternative that can achieve versatile utilization, such as clean heating or electricity at a later stage, on account of the high energy density of hydrogen [5][10][11][12]. Therefore, an increasing number of sustainable pathways for energy conversion and storage technologies, including water electrolysis, batteries, and fuel cells, have been proposed and extensively investigated [9][13][14][15]. Proton exchange membrane water electrolysis (PEMWE) operating in acidic environments has offered an effective way to produce sustainable, high-purity hydrogen through targeted electrochemical reactions since the 1960s [16] (Figure 1). PEMWE has the advantages of a faster dynamic response, a higher current density, and lower crossover of gases and is considered to be the basis of a hydrogen society in the future [17][18][19].

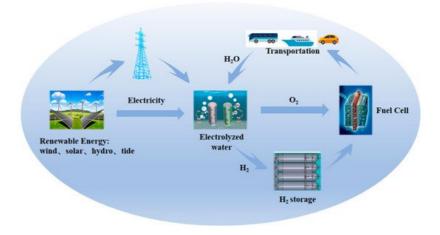


Figure 1. Schematic of sustainable pathways for energy conversion and storage based on electrocatalysis.

Electrochemical water splitting involves two heterogeneous multi-step half-reactions, which are referred to as the cathodic hydrogen evolution reaction (HER) and the anodic oxygen evolution reaction (OER) [20][21]. Owing to the inherent energy barrier, the practical operating voltage of commercial water electrolyzers is higher than the theoretical 1.23 V (versus a reversible hydrogen electrode) under the standard conditions (298 K and 1 atm) [22][23]. For example, industrial electrolytic water generally maintains the external voltage at 1.8~2.0 V [16]. Typically, the descriptor of overpotential is used to show the difference between the thermodynamic potential and the practical potential required to drive the electrochemical reaction [24]. The overpotential mainly comes from the electrochemical polarization on the anode side (η_a) and cathode side (η_c) and the ohmic polarization caused by other resistors (η_{other}) [25]. Comparing η_a and η_{other} , the intrinsically sluggish kinetics of the OER involving a four electron–proton coupled reaction (Equation (1)) hampers the overall water-splitting process [16][26].

$$2H_2O(l) \rightarrow O_2 + 4H^+ + 4e^-$$
 (1)

One solution to this conundrum is to develop suitable catalysts with high efficiency and low overpotential [27]. However, most of the excellent OER catalysts with high activity and durability are not stable in acidic solutions [28]. They are easily oxidized and decomposed in a strong acid system, which is one of the indispensable working conditions for PEMWE [29]. Currently, the iridium (Ir) and ruthenium (Ru)-based electrocatalysts are regarded as the state-of-the-art commercial electrocatalysts for the OER [30][31]. Compared with other catalysts, they exhibit excellent OER catalytic activity due to their inherent promising activity, even if severe corrosion still exists under strong acid working conditions [32]. This provides a driving force for the vast majority of studies on the modifications of these electrocatalysts, including composition, structure, and morphology optimizations [23][33][34][35]. Outstanding OER electrocatalysts should have excellent intrinsic activity and sufficient active sites $\frac{[18]}{}$, and these requirements are generally combined with simplicity and controllability. In this regard, optimizations of the reaction energy barrier, electronic conductivity, and reaction surface area of the OER electrocatalysts are of great importance [18][36][37]. The transport efficiencies of electrons, ions, and produced oxygen are directly related to the number of channels, which depend on rational surface/interface engineering through nanostructural modifications, such as pore size control and construction of a multi-stage structure [33][38]. The nanostructures include zero-dimensional nanoclusters, nanoparticles, nanocages, and nanoframes [39]; one-dimensional nanotubes and nanowires [40]; two-dimensional nanosheets [41]; and three-dimensional nanowire networks, aerogels, etc. [42][43]. Moreover, simple and effective surface/interface engineering techniques have been diversified for adjusting the surface atoms, electronic structures, interfacial stresses, and bridge bonds, such as doping elements, tailoring the coordination environment, and loading with active materials [43][44][45][46].

Although Ru/Ir-based electrocatalysts have indeed shown good OER performance, they are still far from ideal OER electrocatalysts in terms of activity and are not completely stable at high oxidative potentials $^{[17][47]}$. A growing body of evidence shows that Ru-based OER catalysts dissolve extensively during the electrocatalytic process $^{[32]}$. This is because the onset potential of Ru-based catalysts is consistent with the corrosion potential of the metal Ru $^{[48]}$. Ir-based catalysts also suffer similar degradation issues $^{[49]}$. During the long-term catalytic process, rutile oxide of IrO_x will transition to other kinds of phases that are soluble in acid media $^{[50]}$. Therefore, the harsh operating conditions must be taken into account when designing suitable catalysts. Based on this, substantial research efforts have been devoted to investigating the low-precious-metal or precious-metal-free OER catalysts that are stable in acid media, such as perovskite, spinel, and the layer-structure-type family $^{[23][51][52]}$. These kinds of catalysts also exhibit remarkable activity and are low-cost, easily synthesized, and environmentally benign $^{[23]}$.

2. Mechanisms for the OER in Acidic Media

In the case of the OER in acid media, two possible mechanisms built on consecutive proton and electron transfers during the catalytic cycle, known as the adsorbate evolution mechanism (AEM) and the lattice oxygen participation mechanism (LOM), have been widely accepted [24][26][53][54] (Figure 2). For the reaction path based on the AEM, a water molecule first adsorbs on a surface metal cation and decomposes into a proton (H⁺) to form HO*, which further dissociates the second proton to form O* in the second step. After that, HOO* is formed by the nucleophilic attack from another water molecule on the O*. Finally, oxygen is released, accompanied by the desorbed proton. Another four-electron transfer mechanism, known as the LOM, has been proposed based on a series of in-situ isotopic labeling experiments. In contrast to the AEM, lattice O participates in the formation of oxygen for the LOM. Firstly, one water molecule is adsorbed on a surface lattice O and dissociates the first proton to form HO*, which further dissociates the second proton to form O* in the second step. After that, oxygen is released via coupling absorbed O and a surface lattice O along with the presence of a surface oxygen vacancy. Finally, the surface lattice is restored as before via water adsorption and dissociation on the vacancy [24].

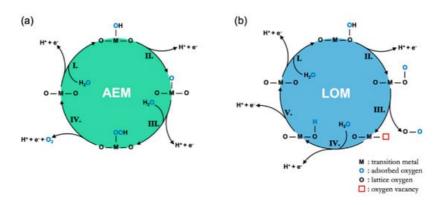


Figure 2. Diagrammatic sketch of **(a)** the AEM mechanism proposed by Rossmeisl et al. and **(b)** the LOER mechanism proposed by Rong et al. [24].

Although both of the mechanisms involve a four-electron transfer, there are still some differences between them [26]. The first one is that the AEM requires a higher reaction energy barrier (>320 mV) than the LOM theoretically [55]. In terms of active sites, the catalytic process predominantly involves a cationic redox (i.e., transition metal ions) in the AEM mechanism and an anion redox (i.e., lattice oxygen) in the LOM [56]. However, it varies according to actual conditions. Jones et al. demonstrated that both mechanisms exist in the OER, which can be detected by the charge storage behavior via the applied bias [57]. At a low bias, it mainly involves the charge storage of metal centers; at a high bias, it involves the storage of oxygen in IrOx. Moreover, strategies for increasing activity are different based on these two different mechanisms. On basis of the AEM, the active metal centers are always at a lower valence state, which can promote the nucleophilic attack of water molecules by increasing the covalence of metal and oxygen [43]. For example, Stoerzinger et al. simulated under-coordinated Ru atoms on a well-crystallized RuO₂ surface with superior OER activity [58]. For the LOM, the metal center is often at a higher valence state, which is committed to promoting the generation of more electrophilic oxygen atoms and increasing the interaction between metal and oxygen [59]. Tarascon et al. studied the lattice oxygen behavior of La₂LiIrO₆. They believed that Ir was not the active site for the OER owing to the pH-dependent activity [60]. Despite the existing difference, some phenomena occurring in the process of an OER can still be explained by these theories. For example, excessive oxidation of metal sites for the AEM and lattice participation for the LOM generally lead to material instability [49]. Furthermore, the reason why the amorphous metal oxides exhibit better catalytic activity is that lattice oxygen can participate in the catalytic reaction easier than the well-crystallized ones [35].

It should be mentioned that, apart from increasing the site density, we can also optimize the composition to modify the intrinsic activity of the OER [33][61]. Recent work has pointed out the superiority of bi-metal oxides as some of the most advanced electrocatalysts toward the OER in acidic media, in terms of features including decreased Ru/Ir contents and enhanced OER activity and selectivity [23][62]. Incorporating suitable foreign metal atoms, such as Cr, Ni, Zn, and Cu atoms, can surprisingly improve the conductivity and alter the electronic structures of the original catalysts, thus enhancing their intrinsic activity $\frac{[63][64]}{}$. It was reported that a low Ru content oxide material ($Cr_{0.6}Ru_{0.4}O_2$) derived from a Cr-based metal-organic framework showed remarkable OER performance in acidic media [65]. The superior catalytic activity and stability can be assigned to the lower occupation at the Fermi level and the altered electronic structures by incorporating Cr. Regarding Ni-modified oxides, it is suggested that Ni serves as the sacrificial component, as its leaching generally leads to enhanced OER activity due to the formation of active OH-containing surface structures [66]. For instance, Ni leaching was observed during the OER process of bulk Ir-Ni mixed oxides with increased Ni contents (21 atomic%, 39 atomic%, and 89 atomic%), and the remaining Ni concentrations were similar in all systems, which turned out to be ~12 atomic% relative to the total amount of Ir and Ni in the oxides [67]. However, differences in their OER performance suggested variations in the resulting active sites caused by the sacrifice of Ni. This means that the OER performance of the catalysts is directly related to the transport kinetics of the electrons involved, and the reaction rate is determined also by the number of active sites. Therefore, it is important to prepare electrocatalysts with a sufficient reaction surface area in order to enable facile mass/electron transport and alter the interaction between metals and supports [68]. The most effective way is to minimize the size of catalyst nanoparticles to within several nanometers to make full use of each active site $\frac{[36]}{}$. In addition, composition modification may also increase the number of catalytic sites. Doping Zn and Cu can confer moderate binding strength on oxygen intermediates, provide more defects or vacancies to enhance the intrinsic activity, and significantly increase the surface area to expose more active reaction sites [69]. However, difficult issues such as well-controlled monodispersity, stabilization of active sites, targeted synthesis, and macro-scale configuration for OER electrocatalysts still remain, especially in acidic media, both experimentally and theoretically [23][70].

3. Tailoring Strategies for Effective OER Electrocatalysts

An important frontier in OER electrocatalysis research is the development of the rational design of catalysts [71]. As discussed above, there are generally two strategies to improve catalytic performance: one is to increase the catalyst's intrinsic activity, and the other is to increase the number of exposed active sites by structure/morphology optimization or by increased loading on a given electrode. Ideally, these two strategies are not mutually exclusive and can be addressed simultaneously, thereby leading to significantly improved activity.

3.1. Metal-Support Interaction

The interface between the metal center and the support will cause the re-arrangement of electrons originating from the support and anchored atoms $\frac{[68][72]}{}$. The re-arranged electrons that have a significant impact on the catalytic performance will be confined in a space several atomic layers thick at the interface $\frac{[73]}{}$. The magnitude and direction of the charge transfer are driven by differences in the Fermi level of the catalytic center and the support $\frac{[50]}{}$. In addition, due to the special microenvironment at the interface, the interface sites will be in direct contact with the catalytic center, the carrier, and the reactants in order to promote the occurrence of synchronous reactions $\frac{[37]}{}$. Additionally, the interface is conducive to the accumulation of excess charge during the charge transfer process, which will strongly promote the catalytic reactions at the interface $\frac{[74]}{}$.

SnO₂ and TiO₂-based compounds are commonly used as supports due to their characteristic of stability $\frac{[75][76]}{[75]}$. In order to improve their low charge-transfer rate, extensive studies have modified them to improve the conductivity. For SnO₂, Sb doping is often used to prepare Sb-SnO₂ (ATO), and the conductivity of the material will be improved due to the increased electron carrier density caused by donor doping $\frac{[77]}{[75]}$. Moreover, the specific surface area and pore volume of SnO₂ can be improved by destroying the long-range order of the original atomic arrangement so as to provide more anchor sites for IrO₂ nanoparticles $\frac{[76]}{[75]}$. The interaction between IrO₂ and the support, the cross-linking morphology of IrO₂, and the porous structure can improve the OER performance of the catalyst. Wang's group designed a kind of Sb-SnO₂ nanowire carrier by an electrospinning method (Figure 3a) $\frac{[37]}{[37]}$. The conductivity can reach 0.83 S·cm⁻¹. Compared with pure IrO₂, the catalytic activity of supported IrO₂/Sb-SnO₂ exhibits significantly improved mass activity, benefiting from the porous structure and the high electronic conductivity of the Sb-SnO₂ support $\frac{[37]}{[37]}$.

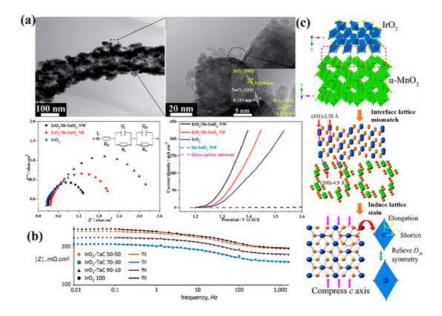


Figure 3. (a) Low- and high-magnification transmission electron microscope (TEM) images, a Nyquist diagram, and the steady-state polarization curve of the $IrO_2/Sb-SnO_2$ catalyst [37]. (b) Bode plot from the electrochemical impedance spectroscopy (EIS) of TaC-supported IrO_2 [78]. (c) Diagram of the process by which the α-MnO₂ substrate induces the lattice strain of IrO_2 [45].

In addition to metal oxides, metal carbides have also emerged as promising OER carriers because of their high conductivity and stability. A TaC-supported IrO_2 catalyst sprayed by Polonsky et al. showed the lowest charge transfer resistance and the highest current density when the loading of IrO_2 reached 70 wt%, which was significantly improved compared with unsupported IrO_2 (Figure 3b) [78]. When TiC is employed as the support for Ir in PEMWE, Ir nanoparticles can be evenly distributed on the TiC surface, and the pore volume of Ir/TiC is twice that of pure Ir. All these advantages make the OER catalytic performance of Ir/TiC much better than that of pure Ir [79].

In general, the existence of a support has two major advantages. On one hand, catalyst particles can be well dispersed on the support surface and facilitate the construction of a three-phase interface consisting of the catalyst, the reactant water molecule, and the produced oxygen $^{[80]}$. In fact, this effect has been widely used to explain the increased activity of supported catalysts. For example, Ir nanoparticles can be well dispersed onto the TiN carrier. The $IrO_2@Ir/TiN$ catalyst prepared by Xing's group showed an enhanced catalytic performance $^{[50]}$. The overpotential was only 265 mV at a current density of 10 mA·cm⁻². Yang et al. synthesized iridium dioxide nanoparticle catalysts with α -MnO₂ nanorods as supports by a simple two-step hydrothermal method. They found that iridium dioxide nanoparticles were subjected to compressive strain due to the lattice mismatch between IrO_2 and α -MnO₂ (Figure 3c) $^{[45]}$.

3.2. Electronic Structure

Incorporation of heteroatoms or groups will destroy the periodicity of the lattice, resulting in the modification of the local coordination environment and the electronic structure of active sites $\frac{[35][81]}{[35][81]}$. This change can effectively regulate the adsorption energy of reaction intermediates and improve the intrinsic activity of electrocatalysts. One of the effective strategies is to incorporate easily soluble non-noble metals, which suffer in-situ dissolution to form an amorphous structure, and increase the degree of coordination unsaturation of the metal in the active center during the oxygen evolution reaction $\frac{[66]}{[66]}$. Zaman et al. chose Ni and Co as the dopants to substitute 50% of the precious metal Ir. The Ni-Co co-doped IrO_2 showed a low overpotential of 285 mV at a current density of 10 mA·cm⁻² (Figure 4a) $\frac{[82]}{[82]}$. Besides Ir-based catalysts, researchers have also done a lot of work on Ru-based catalysts $\frac{[83]}{[83]}$. For example, $SrRuO_3$ exhibited low OER activity in acid electrolytes due to rapid Sr leaching and Ru dissolution at high potential ranges in 0.1 M HClO₄ $\frac{[84]}{[84]}$. Surprisingly, incorporating a small amount of Na^+ into the lattice of $SrRuO_3$ by substituting Sr^{2+} results in significantly enhanced OER performance, both in terms of activity and stability (Figure 4b) $\frac{[85]}{[85]}$. The doped sample exhibited 85% of activity retention after a stability test, which was assigned to the stabilization of Ru centers with a positive shift in dissolution potentials and less distorted RuO_6 octahedra.

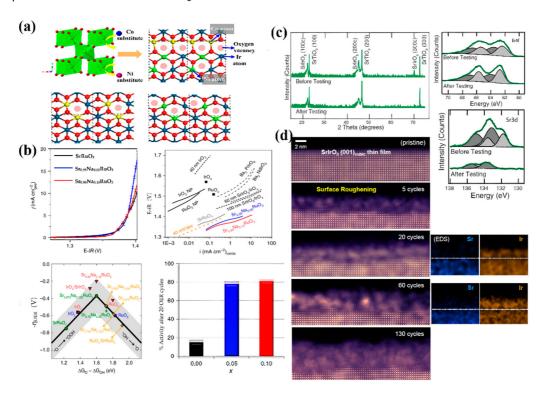


Figure 4. (a) Polyhedral model of IrO_2 being doped with Ni and Co [82]. (b) Enhanced OER activity and durability of $SrRuO_3$ by Na doping [85]. (c) XRD and XPS spectra of a $SrIrO_3$ film before and after 30 h of OER testing [86]. (d) High-angle annular dark-field (HAADF) images of the surface structural evolution of a $SrIrO_3$ film during the OER [87].

Preparation of a perovskite ABO₃ structure (or A₂BB'O₆) and regulation of the valence band structure of B-site cations (usually Ir and Ru) by the atoms at A-site cations have commonly been used to improve the performance of the OER in recent years. Catalysts with this perovskite structure can greatly reduce the usage of noble metals. Pseudocubic SrIrO₃ was the first AIrO₃ single perovskite oxide reported for usage as an OER electrocatalyst in acid media $^{[86]}$. It was found that the formation of IrO_x by Sr leaching on the surface of SrIrO₃ through surface reconstruction was responsible for the enhanced OER activity (Figure 4c). The active surface area increased significantly owing to the formation of IrO_x groups on the surface of catalysts during cycling. Moreover, the intrinsic activity of Ir in the SrCo_{0.9}Ir_{0.1}O₃. δ electrocatalyst was more than two orders of magnitude higher than that of IrO₂ and approximately 10-fold higher as compared with the unmodified benchmark SrIrO₃ δ In observed high activity was attributed to the surface

reconstruction through Sr and Co leaching during the electrochemical cycling, which contains a large number of oxygen vacancies with corner-shared and under-coordinated IrO_x octahedrons in the oxide crystal lattice. Furthermore, surface reconstruction of $AIrO_3$ (A = Sr or Ba) perovskite oxides was visualized by atomic-resolution scanning transmission electron microscopy (Figure 4d) [87]. Leaching of Sr or Ba happened by applying an anodic current, resulting in surface roughening and a structure change by the continuous formation of the mosaic-shaped, Sr-deficient IrO_x on the surface. Moreover, the lattice strain induced by the substitution of smaller lanthanides or yttrium atoms can effectively decrease the adsorption energy toward oxygen-containing intermediates [51]. Although promising in terms of providing enhanced activity, such a surface reconstruction during the electrocatalysis process may be challenged by the uncertainty of the exact structure formed during reactions [86].

It should be mentioned that different kinds of crystal structures have a significant impact on the OER performance. It was proposed that monoclinic $SrIrO_3$ underwent less surface reconstruction than the pseudocubic $SrIrO_3$ owing to the better thermodynamic stability during OER tests in an acidic electrolyte [89]. Strong Ir–Ir metallic bonding and Ir–O covalent bonding in monoclinic $SrIrO_3$ together induced its high structural and compositional stability. Only about 1% of the leached Sr was detected after 30 h in a chronopotentiometry test, which was much less than that (24%) from the pseudocubic $SrIrO_3$. Additionally, Zou et al. put forward another way to decrease the cation leaching and surface reconstruction of pseudocubic $SrIrO_3$ in acid media. They prepared pseudocubic, low-Ir-containing $SrIr_xTi_{1-x}O_3$ perovskite oxides with $0 \le x \le 0.67$. The inert $SrIrO_3$ during $SrIrO_3$ were activated by Ir-substitution and showed remarkable OER activity with a reserved crystal structure of Ir- $SrIrO_3$ during OER cycling $SrIrO_3$.

For metal alloy/oxide-based electrocatalysts, the surface oxidation of metal atoms accompanied by de-alloying (surface dissolution of unstable metals) under acid OER conditions is considered to be a "surface engineering" strategy to design stable and efficient OER electrocatalysts. For example, Travis Jones and Peter Strasser found that electrochemical de-alloying and surface oxidation treatment of $IrNi_{3.2}$ nanoparticle precursors can lead to the dissolution of Ni, and the formed Ni nanoparticles showed better catalytic performance than core—shell-structured $IrNi@IrO_x$ [56]. In order to investigate the relationship between the reconstructed structure and the enhanced OER performance, an operando X-ray absorption spectroscopy (XAS) analysis was performed to characterize the local electronic properties under the OER process. The results show that iridium titania and d-band holes appear in the $IrNiO_x$ electrocatalyst when the potential increases from 0.4 to 1.5 V_{RHE} . More importantly, due to the higher oxidation state of iridium, the iridium oxygen bond length in $IrNiO_x$ was significantly shortened. Based on this unique phenomenon, a structural model of the iridium oxygen ligand environment was proposed. The model shows that the hole-doped iridium ion sites around the electrophilic oxygen ligands form hole-doped IrO_x (caused by Ni leaching) during the OER. Therefore, more electrophilic oxygen ligands are susceptible to the nucleophilic attack of water molecules or hydroxyl ligands, resulting in the formation of oxygen bonds and the reduction of the kinetic energy barrier, which ultimately greatly improves the reaction activity.

3.3. Coordination Environment

Although great efforts have been made to improve the efficiency of OER electrocatalysts, the majority of active sites inside their bulk phases remain inaccessible [23]. In order to maximize the utilization of each active site (approaching 100%) as well as shed light on the effect of the structure of active centers and ligating atoms on the OER activity, catalysts have thus been continuously downsized to the single-atom (SA) level. Single-atom catalysts (SACs) have emerged as a hot new branch of heterogeneous catalysts due to their excellent catalytic performance and financial benefits [28]. Owing to the high requirements for the dispersion, activity, and stability of the atoms, an appropriate support must be selected to optimize the physiochemical properties of the metal atoms anchored [80]. The unique coordination configuration of the dispersed metal atoms and coordinating atoms together create active sites for OER catalysts; therefore, the coordination-activity relationship has a strong impact on the catalytic performance [91]. In fact, targeted synthesis of precious metal single-atom-based OER electrocatalysts remains a bottleneck and has been pursued in the search for better OER catalysis, especially under acid conditions [28]. Generally, carbon supports often suffer severe corrosion problems under acid conditions and oxide supports are not always conductive, which brings about huge trouble for the design of single-atom catalysts [92]. In order to solve these problems, a series of Pt-Cu alloys with an atomically dispersed Ru_1 decoration were studied by Yao et al. [93]. An ultralow overpotential of 170 mV at a current density of 10 mA·cm⁻² was reached by Ru₁-Pt₃Cu in acid media, together with a ten times longer lifetime than a commercial RuO₂ catalyst. Density functional theory calculations suggested that the electronic structure of single Ru sites at the corner or step sites of the Ptrich shell was modulated by the compressive strain in the Pt skin shell, contributing to the optimized binding of oxygen species and improved resistance to over-oxidation and dissolution. Yin et al. demonstrated that surface-exposed Ir single atom couplings with oxygen vacancies anchored in ultrathin NiCo₂O₄ porous nanosheets exhibited remarkable OER activity and stability in acid media, with an overpotential of only 240 mV at a current density of 10 mA·cm⁻² and a longterm durability of 70 h [28]. Based on density functional theory calculations, high electronic exchange and transfer activities

of the surface contributed by Ir atoms anchored at Co sites near oxygen vacancies were determined to be responsible for the prominent OER performance (<u>Figure 5</u>). The synergetic mutual activation between Ir and Co reached the desired H_2O activation level and stabilized *O to boost OER performance.

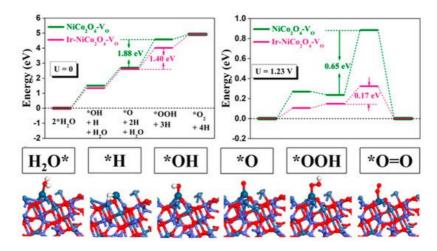


Figure 5. OER pathway under acidic conditions and local structural configurations of intermediates on the $Ir-NiCo_2O_4-V_O$ [28].

3.4. Morphology

The interface between the catalyst and the electrolyte plays an important role in water electrocatalysis $\frac{[94]}{}$. As a very important aspect of a surface structure, morphology has attracted much attention in recent years $\frac{[23][95]}{}$. To date, many methods to control the morphology have been proposed $\frac{[23][61]}{}$.

The morphology of the catalyst can be modified by adopting a suitable preparation strategy, such as the template method, the solvothermal method, or the seed crystal method $\frac{[96][97]}{[97]}$. A nano-porous or ultra-thin structure can increase the number of exposed active sites. Luo et al. prepared a new kind of Ir nanowire with a diameter of 1.7 nm by a wet chemical method (Figure 6a) $\frac{[61]}{[61]}$. Due to the high aspect ratio and large specific surface area, the OER activity increased greatly. The overpotential at 10 mA·cm⁻² in 0.5 M HClO₄ is only 270 mV, which is significantly higher than that of Ir nanoparticles.

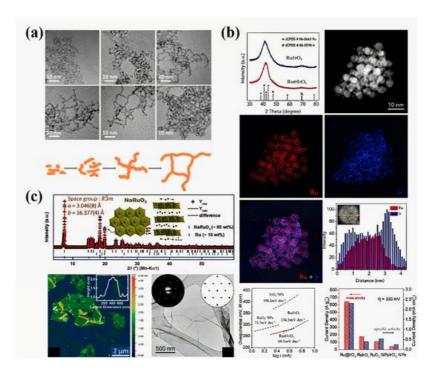


Figure 6. (a) TEM images and a schematic illustration of Ir nanowire intermediates obtained at different reaction times from 5 min to 20 h $^{[61]}$. (b) Fine-structure characterization of a Ru@IrO_x catalyst and investigation of its OER electrocatalytic activity $^{[98]}$. (c) Fine-structure characterizations of ultra-thin NaRuO₂ nanosheets including the crystal structure, an atomic force microscopy (AFM) image, a TEM image, and selected area electron diffraction (SAED) patterns $^{[99]}$

In fact, the electronic structure of catalytic centers can be modified by defects as well as the morphology. For example, Qiao et al. prepared a core–shell Ru@lrO_x icosahedral nanocrystal structure by the sequential polyol method, in which a highly distorted lattice can be observed (Figure 6b) [98]. Due to the interaction between the ruthenium core and the iridium shell, ruthenium is subjected to compressive strain, which was confirmed by the decrease in the distance between ruthenium and ruthenium atoms observed by EXAFS. This strain may lead to a shift in the d-band center, which can regulate the binding energy of oxygen intermediates and the activity of the OER. In addition to the lattice strain, morphology control can also lead to changes in the chemical composition, which directly affects the electronic structure. An example is the ultra-thin ruthenium oxide nanosheets prepared by Lotsch's group (Figure 6c) [99]. Owing to the acid treatment in the stripping process, the actual composition of ruthenium oxide is H_xRuO_2 , in which the valence of ruthenium is +3/+4. Therefore, ruthenium oxide nanoparticles exhibit enhanced OER activity and stability.

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