

Interstitial Atom Engineering

Subjects: Others

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Interstitial light elements play an important role in magnetic materials. Especially, Mn-based compounds with interstitial atoms are important for the easy fabrication of highly functional magnetic devices.

Keywords: interstitial atom ; Mn-based compounds ; rare-earth Fe-based compounds ; rare-earth-based compound ; Bethe-Slater curve ; ferromagnetic ; antiferromagnetic ; permanent magnet

1. Introduction

Some crystal structures possess interstitial crystallographic sites, which light elements such as hydrogen, boron, carbon, nitrogen, and oxygen atoms can occupy. There is a rather long history of metallurgical, physical, and chemical research studies on interstitial atoms [1][2]. In past years, domain control of ferromagnets has been studied [4]. Since the 1980s, interstitial atoms have attracted intense attention related to the improvement of magnetic properties of rare-earth Fe and Co-based permanent magnets [2][3][4][5][6].

Interstitial atoms have two major roles—influencing the stability of the crystal structure and in the modification or change in magnetic properties. In the former case, small amounts of interstitial light elements are required to stabilize the desired crystal structure; compounds without interstitial atoms would not exist in thermal equilibrium. In the latter case, interstitial atoms affect the crystal structure parameters, such as the interatomic distances between magnetic atoms or the orbital hybridization between magnetic and interstitial atoms, consequently meaning the magnetic ordering temperature, the magnetic moment, the magnetic structure, and other factors can be altered.

The most well-studied platforms for interstitial atoms are rare-earth Fe-based permanent magnets. The improvements of the magnetic properties have mainly been achieved through the addition of light elements such as boron, carbon, and nitrogen atoms. The Bethe–Slater curve is one of the criteria needed to understand whether metal 3d transition elements of Cr, Mn, Fe, Co, and Ni possess ferromagnetic (FM) or antiferromagnetic (AFM) states (see Figure 1) [7][8][9][10]. This curve exhibits the exchange coupling as a function of the interatomic distance. Fe falls in the FM region near to the border between FM and AFM states. Therefore, in Fe-based compounds, a shorter Fe–Fe distance (shrinkage of the unit cell volume) favors an AFM state, while a longer Fe–Fe distance (an expansion of the unit cell volume) favors the FM state [11][12][13]. With increasing Fe–Fe distance, smaller overlapping of 3d wave functions makes the 3d band narrower, which leads to the FM state, and in most cases the Curie temperature T_C is enhanced.

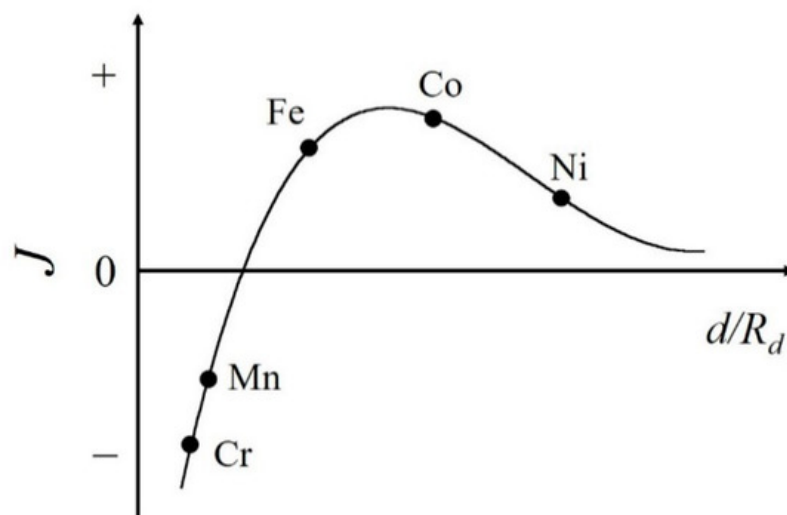


Figure 1. Schematic view of the Bethe–Slater curve. Here, J , d , and R_d represent the magnetic exchange coupling between atoms, the interatomic distance, and the radius of the 3d shell, respectively.

On the other hand, the effects of interstitial atoms in Mn-based compounds are not well researched. As shown in [Figure 1](#), the Mn atom itself shows the AFM ground state, however the expanded Mn–Mn distance in Mn-based compounds leads to the FM state. Mn compounds are indispensable for both FM and AFM materials. For example, MnBi and MnAl have attracted much attention as permanent magnets [\[14\]\[15\]\[16\]\[17\]\[18\]](#). MnSi has been extensively studied as a magnetic material with a skyrmion state, which is a noncollinear magnetic structure. The skyrmion domains can be driven by the low current density threshold [\[19\]](#). Recently, Mn₃Sn has been intensively studied as a topological antiferromagnet [\[20\]](#) and is a candidate next-generation spintronics material. If the effects of interstitial atoms are well understood, they can be highly useful in the development of spintronics devices or highly functional magnetic devices, which can be developed via easy on-demand control of the magnetic state.

2. Mn-Based Compounds

We have surveyed Hydrogen-Absorbed (R or Th)₆Mn₂₃, Hydrogen-Absorbed YMn₂, Carbon-Added Mn₅Si₃, (R or Actinide)Mn₂Si₂ and its germanides, Boron-Added Pd_{0.75}Mn_{0.25} Alloy, and Boron-Added Sm₂Mn₈Al₉B_x. As in the rare-earth Fe-based compounds, the interstitial atoms give rise to the enhancement of FM interaction in the weak hybridization regime leading to the appearance of room temperature ferromagnetism. However, the Mn compounds surveyed above manifest the change or additional formation of magnetism by the interstitial atoms, while many rare-earth Fe-based parent compounds are already ferromagnets. The change from paramagnetic to FM state is observed in hydrogen-absorbed Th₆Mn₂₃, hydrogen-absorbed YMn₂ or Sm₂Mn₈Al₉B_x. The result of Mn₅Si₃C_x thin film may be a rare example of change from the AFM to FM state by the interstitial atoms. In Pd_{0.75}Mn_{0.25}B_x, the room temperature ferromagnetism is induced by a slight addition of boron, while the low-temperature magnetic ground state of the parent compound is unchanged. This can be regarded as an example of the additional formation of magnetism by interstitial atoms. It should be noted that, in some cases, the change or additional formation of a magnetic state seems to abruptly occur, which is valuable for future research. We note here that the magnetic structures have been divided into FM and AFM, although some compounds may show a more complicated state such as canted AFM, and spiral AFM. In the future, discussion taking into account a more microscopic mechanism of the magnetic ordering would be necessary.

3. Perspectives

3.1. Application of Mn-Based Magnetic Materials

The change in magnetism between FM and AFM states or the additional formation of new magnetic coupling at rather high temperatures by interstitial atoms is substantially valuable in a magnetic device integrated with both FM and AFM materials due to the easy on-demand control of magnetism in the fabrication process. At the present stage, Mn-based compounds fulfill the requirement of change or the additional formation of magnetism, while in Fe-based compounds, only improvements of FM properties are extensively investigated and the change (or the additional formation) of the magnetic state is not well explored. Focusing on the research area of the permanent magnet, a rare-earth Mn-based permanent magnet is still missing, although the MnBi-type magnets are well known. Based on the Bethe–Slater curve, Mn atoms favor the FM state with expanding Mn–Mn distance. Therefore, the density of Mn could not be so increased as in rare-earth Fe-based permanent magnets, resulting in a smaller saturation magnetization. However, there exists a large gap of *BH* energy product between the NdFeB magnets and ferrite magnets, and a rare-earth Mn-based permanent magnet may be a good candidate filling the gap [\[21\]](#).

3.2. Towards Further on-Demand Control of Magnetism

Further improvement of magnetic properties in the on-demand control would be achieved by another strategy such as a composition effect and carrier doping. If a metallurgical phase diagram of a target compound possesses a homogeneity range, the magnetic ordering temperature often varies with the atomic composition. For example, the *T_C* of Tb₂Co₂Ga ranges rather widely from 75 to 145 K by changing the starting composition [\[22\]](#). Such a composition effect is reported in other compounds [\[23\]\[24\]\[25\]](#) such as Nd₃Pd₂₀Ge₆, Tb₃Co₃Ga and Mn_{1+x}Ga. In many cases, the crystal structure parameters slightly change, which heavily affects the magnetic exchange interactions.

The magnetic anisotropy energy is one of the important factors in characterizing a ferromagnet. It is known that it can be tuned by doping, mainly due to the variation of the density of states near the Fermi level. The doping effect is reported in, for example, Ni₂MnGa, SmCo_{5-x}Fe, MnBi, Nd₂Fe₁₇X (X = C or N), Ce₂AuP₃ and so on [\[26\]\[27\]\[28\]\[29\]\[30\]](#). Taking into account the crystal symmetry, which is related to the magnetic anisotropy energy, a lower crystal symmetry with more tunable crystal parameters might be favorable.

3.3. Comments on Control of Magnetism by External Field

Another interesting control of magnetism is the manipulation of spin by external fields such as the electric field and the optical light. For example, the employment of an electric double-layer transistor has achieved a control of magnetism by weak voltage [31]. Recently, an optical change in magnetism through the Kondo effect has been reported [32]. The interstitial atoms may precisely control the magnetism so that only a small magnitude of external field is required for the device working, and the power consumption can be highly reduced.

3.4. Comments on Critical Behavior

From the fundamental viewpoint, research into critical behavior is interesting. Actually, in strongly correlated electron systems, there have been plenty of studies for seeking a quantum critical point under the suppression of magnetism [33][34][35][36][37]. We speculate that a formation of FM exchange coupling above room temperature would be a discontinuous phenomenon as mentioned in the results of Mn-based compounds. While it is not well investigated for rare-earth Fe-based compounds, we note that $\text{RFe}_{11}\text{TiC}_x$ and $\text{RFe}_{11}\text{TiC}_x$ show a finite change in T_C at an infinitely zero value of volume expansion depending on R species [38].

References

1. Goldschmidt, H.J. Interstitial Alloys; Butterworths: London, UK, 1967.
2. Grandjean, F.; Long, G.J.; Buschow, K.H.J. Interstitial Intermetallic Alloys; Kluwer Academic Publishers: Dordrecht, The Netherlands, 1995.
3. Yang, Y.; Yang, J.; Han, J.; Wang, C.; Liu, S.; Du, H. Research and Development of Interstitial Compounds. *IEEE Trans. Magn.* 2015, 51, 2103806.
4. Nieves, P.; Arapan, S.; Maudes-Raedo, J.; Marticorena-Sánchez, R.; Del Brío, N.L.; Kovacs, A.; Echevarria-Bonet, C.; Salazar, D.; Weischenberg, J.; Zhang, H.; et al. Database of novel magnetic materials for high-performance permanent magnet development. *Comput. Mater. Sci.* 2019, 168, 188–202.
5. Coey, J.M.D. Perspective and Prospects for Rare Earth Permanent Magnets. *Engineering* 2020, 6, 119–131.
6. Hadjipanayis, G.C.; Gabay, A.M.; Schönhöbel, A.M.; Martín-Cid, A.; Barandiaran, J.M.; Niarchos, D. ThMn12-Type Alloys for Permanent Magnets. *Engineering* 2020, 6, 141–147.
7. Slater, J.C. Cohesion in Monovalent Metals. *Phys. Rev.* 1930, 35, 509–529.
8. Slater, J.C. Atomic Shielding Constants. *Phys. Rev.* 1930, 36, 57–64.
9. Sommerfeld, A.; Bethe, H. *Handbuch der Physik*; Springer: Berlin, Germany, 1933; Volume 24, p. 595.
10. Cardias, R.; Szilva, A.; Bergman, A.; Marco, I.D.; Katsnelson, M.I.; Lichtenstein, A.I.; Nordström, L.; Klautau, A.B.; Eriksson, O.; Kvashnin, Y.O. The Bethe-Slater curve revisited; new insights from electronic structure theory. *Sci. Rep.* 2017, 7, 4058.
11. Givord, D.; Lemaire, R. Magnetic Transition and Anomalous Thermal Expansion in R_2Fe_{17} Compounds. *IEEE Trans. Magn.* 1970, MAG-10, 109–113.
12. Palstra, T.T.M.; Nieuwenhuys, G.J.; Mydosh, J.A.; Buschow, K.H.J. Mictomagnetic, ferromagnetic, and antiferromagnetic transitions in $\text{La}(\text{Fe}_x\text{Al}_{1-x})_{13}$ intermetallic compounds. *Phys. Rev. B* 1985, 31, 4622–4632.
13. Kitagawa, J.; Yakabe, G.; Nakayama, A.; Nishizaki, T.; Tsubota, M. Competition between ferromagnetic and antiferromagnetic states in $\text{Al}_{8.5-x}\text{Fe}_{23}\text{Ge}_{12.5+x}$ ($0 \leq x \leq 3$). *J. Solid State Chem.* 2020, 284, 121188.
14. Chen, D.; Gondō, Y. Temperature Dependence of the Magneto-Optic Effect and Resonance Phenomena in Oriented MnBi Films. *J. Appl. Phys.* 1964, 35, 1024–1025.
15. Park, J.H.; Hong, Y.K.; Bae, S.; Lee, J.J.; Jalli, J.; Abo, G.S.; Neveu, N.; Kim, S.G.; Choi, C.J.; Lee, J.G. Saturation magnetization and crystalline anisotropy calculations for MnAl permanent magnet. *J. Appl. Phys.* 2010, 107, 09A731.
16. Gabay, A.M.; Hadjipanayis, G.C.; Cui, J. New anisotropic MnBi permanent magnets by field-annealing of compacted melt-spun alloys modified with Mg and Sb. *J. Magn. Mater.* 2020, 495, 165860.
17. Xiang, Z.; Xu, C.; Wang, T.; Song, Y.; Yang, H.; Lu, W. Enhanced magnetization and energy product in isotropic nanocrystalline $\text{Mn}_{55}\text{Al}_{45}$ alloys with boron doping. *Intermetallics* 2018, 101, 13–17.
18. Zhao, S.; Wu, Y.; Zhang, C.; Wang, J.; Fu, Z.; Zhang, R.; Jiang, C. Stabilization of τ -phase in carbon-doped MnAl magnetic alloys. *J. Alloy. Compd.* 2018, 755, 257–264.

19. Jonietz, F.; Mühlbauer, S.; Pfleiderer, C.; Neubauer, A.; Münzer, W.; Bauer, A.; Adams, T.; Georgii, R.; Böni, P.; Duine, R.A.; et al. Spin Transfer Torques in MnSi at Ultralow Current Densities. *Science* 2010, 330, 1648–1651.
20. Nakatsuji, S.; Kiyohara, N.; Higo, T. Large anomalous Hall effect in a non-collinear antiferromagnet at room temperature. *Nature* 2015, 527, 212–215.
21. Coey, J.M.D. New permanent magnets; manganese compounds. *J. Phys. Condens. Matter* 2014, 26, 064211.
22. Tanaka, S.; Terada, H.; Shirakawa, N.; Tsubota, M.; Kitagawa, J. The Impact of the Composition Effect on Ferromagnetic Properties of Tb₂Co₂Ga. *Metals* 2019, 9, 1242.
23. Kitagawa, J.; Takeda, N.; Sakai, F.; Ishikawa, M. Effect of composition in (RE)₃Pd₂₀Ge₆ (RE = La, Ce and Nd). *J. Phys. Soc. Jpn.* 1999, 68, 3413–3416.
24. Kitagawa, J.; Terada, H.; Shirakawa, N.; Tsubota, M.; Nose, A.; Tanaka, S. Composition effect in ferromagnetic properties of Tb₃Co₃Ga. *Results Phys.* 2019, 15, 102591.
25. Lu, Q.M.; Yue, M.; Zhang, H.G.; Wang, M.L.; Yu, F.; Huang, Q.Z.; Ryan, D.H.; Altounian, Z. Intrinsic magnetic properties of single-phase Mn_{1+x}Ga (0 < x < 1) alloys. *Sci. Rep.* 2015, 5, 17086.
26. Enkovaara, J.; Heczko, O.; Ayuela, A.; Nieminen, R.M. Coexistence of ferromagnetic and antiferromagnetic order in Mn-doped Ni₂MnGa. *Phys. Rev. B* 2003, 67, 212405.
27. Larson, P.; Mazin, I.I.; Papaconstantopoulos, D.A. Effects of doping on the magnetic anisotropy energy in SmCo₅-xFex and YCo₅-xFex. *Phys. Rev. B* 2004, 69, 134408.
28. Sakuma, A.; Manabe, Y.; Kota, Y. First principles calculation of magnetocrystalline anisotropy energy of MnBi and MnBi_{1-x}Sn_x. *J. Phys. Soc. Jpn.* 2013, 82, 073704.
29. Pandey, T.; Parker, D.S. Magnetic properties and magnetocrystalline anisotropy of Nd₂Fe₁₇, Nd₂Fe₁₇X₃, and related compounds. *Sci. Rep.* 2018, 8, 3601.
30. Miyahara, J.; Shirakawa, N.; Setoguchi, Y.; Tsubota, M.; Kuroiwa, K.; Kitagawa, J. Hill plot focusing on Ce compounds with high magnetic ordering temperatures and consequent study of Ce₂AuP₃. *J. Supercond. Nov. Magn.* 2018, 31, 3559–3564.
31. Yamada, Y.; Ueno, K.; Fukumura, T.; Yuan, H.T.; Shimotani, H.; Iwasa, Y.; Gu, L.; Tsukimoto, S.; Ikuhara, Y.; Kawasaki, M. Electrically Induced Ferromagnetism at Room Temperature in Cobalt-Doped Titanium Dioxide. *Science* 2011, 332, 1065–1067.
32. Kitagawa, J.; Kitajima, D.; Shimokawa, K.; Takaki, H. Photoinduced Kondo effect in CeZn₃P₃. *Phys. Rev. B* 2016, 93, 035122.
33. Stewart, G.R. Non-Fermi-liquid behavior in d- and f-electron metals. *Rev. Mod. Phys.* 2001, 73, 797–855.
34. Eom, D.; Ishikawa, M.; Kitagawa, J.; Takeda, N. Suppression of Antiferromagnetism by Kondo Effect and Quantum Critical Behavior in CeCoGe 3-xSix (0 ≤ x ≤ 3). *J. Phys. Soc. Jpn.* 1998, 67, 2495–2500.
35. Custers, J.; Gegenwart, P.; Wilhelm, H.; Neumaier, K.; Tokiwa, Y.; Trovarelli, O.; Geibel, C.; Steglich, F.; Pépin, C.; Coleman, P. The break-up of heavy electrons at a quantum critical point. *Nature* 2003, 424, 524–527.
36. Löhneysen, H.V.; Rosch, A.; Vojta, M.; Wölfle, P. Fermi-liquid instabilities at magnetic quantum phase transitions. *Rev. Mod. Phys.* 2007, 79, 1015–1075.
37. Huang, C.-L.; Hallas, A.M.; Grube, K.; Kuntz, S.; Spieß, B.; Bayliff, K.; Besara, T.; Siegrist, T.; Cai, Y.; Beare, J.; et al. Quantum Critical Point in the Itinerant Ferromagnet Ni_{1-x}Rh_x. *Phys. Rev. Lett.* 2020, 124, 117203.
38. Akayama, M.; Fujii, H.; Yamamoto, K.; Tatami, K. Physical properties of nitrogenated RFe₁₁Ti intermetallic compounds (R = Ce, Pr and Nd) with ThMn₁₂-type structure. *J. Magn. Magn. Mater.* 1994, 130, 99–107.