# **Polyaryletherketone Based Blends**

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Polyaryletherketone-based thermoplastic blends (PAEK) are high-performance copolymers able to replace metals in many applications including those related to the environmental and energy transition. PAEK lead to the extension of high-performance multifunctional materials to target embedded electronics, robotics, aerospace, medical devices and prostheses. Blending PAEK with other thermostable thermoplastic polymers is a viable option to obtain materials with new affordable properties.

Keywords: polymer blend ; miscibility ; thermal transition ; crystallization

## 1. Introduction

High-performance thermoplastic (HPT) polymers lead the extension of advanced tailored applications such as embedded electronics, robotics, aerospace, medical devices and prostheses. In addition, HPT research and growth are driven by their use in structural composite materials, in which thermoplastics are an option for the matrix, offering significant weight savings and time-to-market reduction compared to thermoset composites or incumbent steel. Other widespread applications of HPT are as membranes from the barrier, filtration, osmosis and as templates for chemical reactions, such as proton exchange membrane fuel cells.

All PAEKs offer a compromise between thermal stability, mechanical properties, chemical resistance and durability. The enhancement of these features can lead to the development of cutting-edge applications <sup>[5]</sup>. Ideally, one material should fit all the targeted properties, namely "multi-functional" material. Blending two polymers may be the easiest option to design such new materials. Indeed, two or more thermoplastics, often chosen such as their properties complement one another, yield a new material with synergetic properties.

## 2. Presentation of Blends Components

In this section, the polymer blends are classified according to their miscibility and then the chemical structures and properties of PAEKs and other high-performance thermoplastics are briefly presented.

### 2.1. Classes of Blends

Many attempts have been constructed to classify the polymer blends: miscible, immiscible, compatible, incompatible, partially miscible, etc. The goal here is not to review terminology or to discuss which definition is best suited to each situation. This text will use the following terms to classify the PAEK/HPT blends: miscible, partially miscible and immiscible. Miscible blends behave in a single phase and exhibit a single glass transition temperature. Partially miscible blends exhibit miscibility to an extent, as a function of blend composition or process parameters. It is worth mentioning that, whatever the polymer added to PAEK, the miscibility depends on the molecular mass. However, this information is

barely provided in the reviewed articles. Finally, the blends that do not fall into the previous categories are classified as immiscible blends. **Table 1** shows the common PAEK/HPT blends found in the literature. It is noteworthy that PEI is the only HPT that forms a fully miscible blend with various PAEK in the melted state. However, PEI comes in different conformations, and some of them are not miscible with PAEK.

 Table 1. Classification of the most common HPT blended with PAEK as a function of their miscibility and method of blending.

Blending Method	Miscible Blends	Partially Miscible Blends	Immiscible Blends
Melt-mixing	PAEK <i>Imeta</i> -PEI	PAEK/PAI	PEEK/para-PEI
		PAEK/PES	PEEK/ortho-PEI
		ΡΕΚΚ/ΤΡΙ ΡΕΚ/ΤΡΙ	PEEK/TPI
		PAEK/LCP	PEEK/PBI
			PAEK/PTFE
Solution blending	SPEEK/PAI	PEKK/PBI	
	SPEEK/PEI	PEEK/PES	
	SPEEK/PBI		
	SPEEK/PES		

#### 2.2. Presentation of PAEK

Polyaryletherketones are semicrystalline high-performance thermoplastics with strong molecular rigidity of their repeating units. They demonstrate high-temperature stability, chemical resistance and high mechanical strength over a wide temperature range. PEEK (**Figure 1**), PEKK (**Figure 2**), PEK (**Figure 3**) and others have similar crystalline structures of two-chain orthorhombic packing <sup>[6]</sup> but not the same ether/ketone content <sup>[7]</sup>.



Figure 1. Structural formula of PEEK.



Figure 2. Chemical structure of PEKK.



Figure 3. Scheme of PEK unit.

Polyetheretherketone combines the strength of 98 MPa and the stiffness of 125 MPa with a very good tensile fatigue of 97 MPa, thermal and chemical resistance (including the majority of organic solvents, oils and acids). Its mechanical properties remain stable up to temperatures of about 240 °C [8].

Polyetherketone is a polymer with high heat resistance above 300 °C, chemical resistance and an ability to withstand high mechanical loads from 88 MPa to 112 MPa  $\frac{9[10][11]}{10}$ . This polymer is synthesized in various formulations with individually unique properties. The PEKK formulations are expressed by the ratio of the percent of terephthaloyl (T) to isophthaloyl (I) moieties used during the synthesis that created the polymer. The T/I ratio affects the melting point ranging from 305 °C to 360 °C, the glass transition temperature from 160 °C to 165 °C and the crystallization kinetics  $\frac{120}{100}$ .

Polyetherketone is characterized by very good material properties such as a tensile strength of 110 MPa, an elastic modulus of 4200 MPa and an elongation at break of 35%. It has a high resistance to abrasion and increased compression strength of 180 MPa at higher temperatures <sup>[13]</sup>.

The properties of polyaryletherketones are due to the occurrence of phenylene rings linked via oxygen bridges (ether, R– O–R) and carbonyl groups (ketone, R-CO-R) in different configurations and proportions. The glass transition temperature, T<sub>g</sub>, and the melting temperature, T<sub>m</sub>, of the polymer depend on the ratio and sequence of ethers and ketones. They also affect its heat resistance and processing temperature. The lower the ratio of ether/ketone, the more rigid the polymer chain is and the higher the T<sub>g</sub> and T<sub>m</sub> as seen in **Figure 4**. As a consequence, the processing temperature ranges from 350 °C to 400 °C <sup>[14]</sup>. The main suppliers are Victrex, Arkema, Solvay, Evonik, Sabic and Gharda.



Figure 4. The melting temperature and glass transition temperature of PAEK as a function of ether/ketone ratio [15].

Another type of PAEK was launched recently on the market by Victrex, with a lower melting temperature while keeping the same glass transition, maned as LM-PAEK (LM: low melting). The typical trend for  $T_g$  and  $T_m$  as seen in **Figure 4** is not valid anymore, due to the higher rigidity of its monomers. Furthermore, to increase the hydrophilicity of PEEK materials, charged groups are introduced using sulfuric acid into the polymer chains to make them ion-exchangeable. Sulfonation aids in the transport of cations and increases the hydrophilicity of the polymer. Sulfonated PEEK is presented in **Figure 5**.



Figure 5. Sulfonated PEEK structure.

#### 2.3. Presentation of HPT Usually Blended with PAEK

Polyetherimide is an amorphous polymer whose chemical structure of grade Ultem 1000 from Sabic is seen in **Figure 6**. It is renowned for being inherently flame retardant. Ultem 1000 resin in an unreinforced general-purpose grade offers a high strength of 105 MPa and a modulus of 3.2 GPa and a broad chemical resistance up to high temperatures of 170 °C while maintaining stable electrical properties over a wide range of frequencies <sup>[16][17]</sup>. The ketone groups in its backbone render it more flexible than polyimides, hence better processability. Similar to other amorphous thermoplastics, its mechanical

strength decreases fast above its T<sub>g</sub> at 215 °C. PAEK/PEI blends have been reported to be miscible, especially the binary blend PEEK/PEI. The latter is commonly used in PEEK composite parts <sup>[18][19][20]</sup>, where the PEI is a joining agent, a method referred to as "Thermabond" <sup>[21]</sup>. Also, PEI is used as an energy director, meaning interfacial film, to assemble carbon fiber/PAEK composites through ultrasonic welding <sup>[22]</sup>. PEEK/PEI blends have also found application as tribological material <sup>[23]</sup>, biomedical implants <sup>[24]</sup> and lightweight foam structures <sup>[25]</sup>. Since this blend has been reported to be miscible at all compositions in the amorphous state, it has attracted considerable attention to further the fundamental understanding of miscible blends <sup>[26][27][28][29][30]</sup>.



Figure 6. Schematic structure of PEI Ultem 1000.

Most studies with polyamideimides mention Torlon<sup>®</sup> 4000T from Solvay, whose chemical structure is presented in **Figure 7**, as well as other PAI Torlon<sup>®</sup>. Torlon<sup>®</sup> 4000T is the unfilled PAI powder mainly for adhesive applications <sup>[31]</sup>. Some authors have synthesized their PAI <sup>[32]</sup> with specific properties. Their mechanical, thermal and oxidative properties make them suitable for various applications thanks to a high glass transition temperature of around 275 °C <sup>[33]</sup>.



Figure 7. Chemical structure of the PAI Torlon<sup>®</sup> 4000T.

The chemical structure of polybenzimidazole is presented in **Figure 8**. Its rigid structure gives excellent thermal and mechanical properties, such as a melting point above 600 °C <sup>[34]</sup> and a glass transition of 435 °C <sup>[35]</sup>. It has the highest tensile strength among high-performance polymers, up to 145 MPa, and offers good chemical resistance <sup>[34][36][37][38]</sup>. However, it has a large water uptake of 15 wt.%, and the imidazole ring in the repeat unit may be subjected to hydrolysis, reducing its lifetime in applications such as fuel cells <sup>[39]</sup>.



Figure 8. Scheme of a PBI unit.

Polyethersulfone, whose chemical structure is sketched in **Figure 9**, is an amorphous thermoplastic with a  $T_g$  between 190 °C and 230 °C, yielding high thermal stability. It also offers high mechanical rigidity and creep resistance, which make it a good candidate to blend with PAEK <sup>[40]</sup>.



Figure 9. The repeat unit of a PES.

Thermoplastic polyimides, whose structure is depicted in **Figure 10**, display high mechanical properties of 128.7 MPa at tensile strength and 14.2% at elongation at break for TPI-4 <sup>[41]</sup>. They have one of the highest continuous operating temperatures for an unfiled thermoplastic of 240 °C <sup>[42]</sup>. Their T<sub>g</sub> is at 250 °C. The main drawbacks of TPI are their high melt viscosity and low chemical resistance, making their processing challenging. Modification of the aromatic backbone by the inclusion of flexible functional bonds has enabled the synthesis of polyimide with a lower melt viscosity <sup>[43][44]</sup>. While TPIs are amorphous (Matrimid 5218, LaRC-TPI, Extem XH/UH), semicrystalline TPIs (with a crystallinity at 20%) were developed as an alternative, such as the so-called "New-TPI" by Mitsui Toatsu Chemical, Inc. (Tokyo, Japan) <sup>[45]</sup>, later renamed "Aurum" or "Regulus" <sup>[46]</sup> depending on its end-use. Both amorphous and semicrystalline TPI suffer from low processability, even though the N-TPI has better overall mechanical properties and chemical resistance than TPI. Blending TPI or N-TPI with PAEK is an interesting way to improve the processability of the former, despite their immiscibility with PAEK.



Figure 10. TPI repeat unit.

Polytetrafluoroethylene is a synthetic fluorocarbon with a high molecular weight compound consisting of carbon and fluorine. Their structure is shown in **Figure 11**. It is highly hydrophobic, biocompatible and widely used as a solid lubricant thanks to its low friction coefficient. Concerning engineering applications, PEEK is wear-resistant but may suffer from wear loss or high friction coefficient at elevated temperatures. Blending PEEK with PTFE is a way to improve the tribological performances of PAEK-based blends. But PTFE is extremely viscous in the melted state, which complicates its processability. Its glass transition temperature is around 114 °C and its melting point is 320 °C and it starts deteriorating above 260 °C, making it difficult to blend with PAEK.



Figure 11. Schematic structure of the PTFE unit.

Liquid crystal polymers are characterized by the properties of liquid crystal, which can be those of conventional liquids or those of solid crystals. For example, a liquid crystal may flow like a liquid, but its molecules may be oriented in a crystal-like way. Those thermoplastics showed up on the market in the 1980s. Thermotropic LCP, shown in **Figure 12**, is melt processable, while lyotropic LCP with a higher chain rigidity and intermolecular bonding is spun from a solution such as sulfuric acid. During melt processing, the rigid molecules with units disrupt chain linearity and pack slightly to reduce the melting point. It has been reported that the blends with LCP often tend to be immiscible but have a range of useful properties. TLCP have been blended with many thermoplastics including PEEK <sup>[47]</sup>.



Figure 12. Repeat unit of TLCP.

## 3. Thermodynamics of Miscibility and Morphologies

Polymer–polymer theory of miscibility is based on the Flory–Huggins theory for a solution, for which, ideally, each part of the solute fills a space in the solution that it is soluble in, e.g., the so-called "lattice model" <sup>[48]</sup>. Polymers, and especially HPT, are at best semirigid due to their aromatic backbone. The Flory–Huggins theory still serves to describe the polymer–polymer mixture. The free energy of mixing depends on combinatorial entropy since the polymer of the mixture cannot fill all the space offered by the lattice due to its restricted degree of freedom. This effect is taken into account by considering

the volume of the polymer, i.e., steric effects <sup>[49]</sup>. The free energy of mixing depends also on the interaction between like and unlike pairs of monomers, which is an enthalpic contribution. Usually, since macromolecules contain a large number of monomers, the repulsion between macromolecules is large <sup>[50]</sup>. The strength of this interaction, as the difference between like and unlike pairs, is defined as the interaction Flory–Huggins parameter,  $\chi$ . The free energy of mixing is usually positive, i.e., unfavorable to mixing. However, strong specific interactions (dipole–dipole, ion–dipole, hydrogen bonding, acid-base or charge transfer) <sup>[51]</sup> may occur, and these will counterbalance the unfavorable entropic (steric) and enthalpic (van der Walls) interactions to give miscibility. Additionally, the free energy of mixing must have a positive curvature (in other words, increased differential growth at the center, for example, the surface of the sphere) at any concentration to give a miscible blend concerning the negative enthalpic term. Usually, this is accomplished by specific interactions, such as dipole–dipole and hydrogen bonding <sup>[52]</sup>. Such interactions are highly dependent on temperature; as a consequence, the miscibility depends on the temperature. This dependency is evidenced by miscibility diagrams depicting an upper critical solution temperature (UCST) or a lower critical solution temperature (LCST).

Partially miscible blends have a negative curvature (in other words, increased differential growth at the edges, for example, "saddle") for some concentrations on a miscibility diagram <sup>[53]</sup>. For example, PEEK/PES blends are partially miscible; their miscibility depends on the processing method. A probable cause, in view of the discussion above, would be that their respective conformation does not allow the macromolecules to interact attractively. However, sulfonated PEEK is fully miscible with PES, mainly due to the presence of sulfonate groups, which would specifically interact with the likes on the PES backbone. Another example would be PEEK/PBI blends, where steric repulsion would be the main reason for their immiscibility. Nevertheless, SPEEK is fully miscible with PBI, mainly due to the acid-base interaction between the basic, amide groups of the PBI and the acid, sulfonate groups of the SPEEK.

When functionalization is not possible, immiscible polymers may be compatibilized. The addition of a compatibilizer is the main route to obtain melt-mixed polymer blends <sup>[54][55]</sup>. The compatibilizers may be classified into two categories:

- Nanoparticles, which are partially or fully miscible with the polymers in the blend <sup>[56]</sup>. Usually, they are amphiphilic, "Janus" nanoparticles <sup>[57]</sup> aimed at the interface to reinforce it and promote the stress transfer between the two phases;
- Block copolymers: For instance, PET/PP blends are often compatibilized by PP-b-MA (MA: maleic anhydride) copolymers, where the *block* MA is miscible with PET. Gao et al. <sup>[58]</sup> compatibilized PEEK/PI blends with PEEK-b-PI block copolymers. In any case, the length of the blocks and the number of copolymers control the stability and the final morphology of the blends. It is worth noting that the choice of copolymers available to compatibilize PAEK with another HPT is very narrow.

Immiscible blends exhibit a two-phase morphology, where each phase is a rich phase of one component, whereas miscible blends should behave as a single-phase material. Compatibilization leads to fine dispersion of one of the phases into the other, depending on the composition. Partially miscible blends have a spectrum of morphologies depending on the composition and processing parameters, as in **Figure 13**. Of interest is obtaining a co-continuous structure <sup>[59]</sup>. Indeed, even if the blend has an immiscible state, a co-continuous morphology may give a blend with mechanical properties better than expected. Thus, if blending two polymers leads to an immiscible blend, but they have complementary properties (for instance, thermal stability or solvent resistance), one may want to process them with a set that gives a co-continuous morphology to obtain enhanced properties. Such morphologies are expected when blending PAEK with other thermoplastics, likewise with conventional polymer blends.



Figure 13. Basic types of phase structures in polymer blends.

Moreover, due to the crystallization of PAEK, the resulting morphology depends on the crystallinity generated in the blend. In the PAEK/HPT blends, the HPT are mostly amorphous, so the semicrystalline/amorphous blends may develop three main configurations [27][28][60][61].

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