

Metal Oxide-Based Photocatalysts for CO₂ Reduction

Subjects: [Chemistry](#), [Analytical](#)

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The photoconversion of CO₂ into solar fuels seems to curb greenhouse effect and resolve the energy crisis. One of the challenges in developing practical CO₂ photoconversion catalysts is to design materials with a low cost, high activity and good stability. The excellent photocatalysts based on TiO₂, WO₃, ZnO, Cu₂O and CeO₂ metal oxides that are cost-effective and long-lasting were discussed. Strategies to improve CO₂ photoconversion efficiency are summarized and photocatalysts forms of 0D, 1D, 2D and 3DOM (zero/one/two-dimensional and three-dimensional-ordered macroporous, respectively) are involved, which can inspire the future improvement in photochemistry.

photochemistry

CO₂ reduction

metal oxide materials

solar fuels

1. Introduction

The transformation of CO₂ can be driven by illumination, electricity and heat [\[1\]](#). Solar energy is a safe, clean, renewable and inexhaustible energy source, so it is ingenious to achieve this conversion with sunlight [\[2\]](#). Additionally, the reaction conditions in photocatalytic processes are mild [\[3\]](#); thus, it is easy to conduct experimental tests. A new world opened up since H₂ and O₂ were obtained after radiating TiO₂ with light. Extensive research about photocatalyst were conducted until the year of 2000. Since then, a growing number of materials have been designed and studied to absorb solar energy, including oxide semiconductors, sulfides (ZnS, CdS and MoS₂), and polyoxometalates (Bi₂WO₆, Bi₂MoO₆ and BiFeO₃) [\[4\]\[5\]](#). Organics, organometallic complexes, covalent organic polymers and noncovalent self-assembled supramolecular organic matter are also involved. Among them, inorganic metal oxide materials are widely studied for establishing efficient artificial photosystems due to their low cost, facile synthesis, stable crystal structures and environmental friendliness. TiO₂, Cu₂O, ZnO, WO₃ and CeO₂ show promising research value as the most common materials.

2. Theoretical Foundation and Strategies of Photocatalytic CO₂ Reduction

The photocatalytic process in semiconductors can generally be described as shown in **Figure 1**. Upon being excited by an incident photon with energy equal to or higher than the bandgap (E_g), charge carriers are generated. Electrons (e^-) at the bottom of conduction band (CB) migrate to the surface of the catalyst to initiate reduction reactions with CO₂. Holes (h^+) at the top of valence band (VB) conduct oxidative reactions.

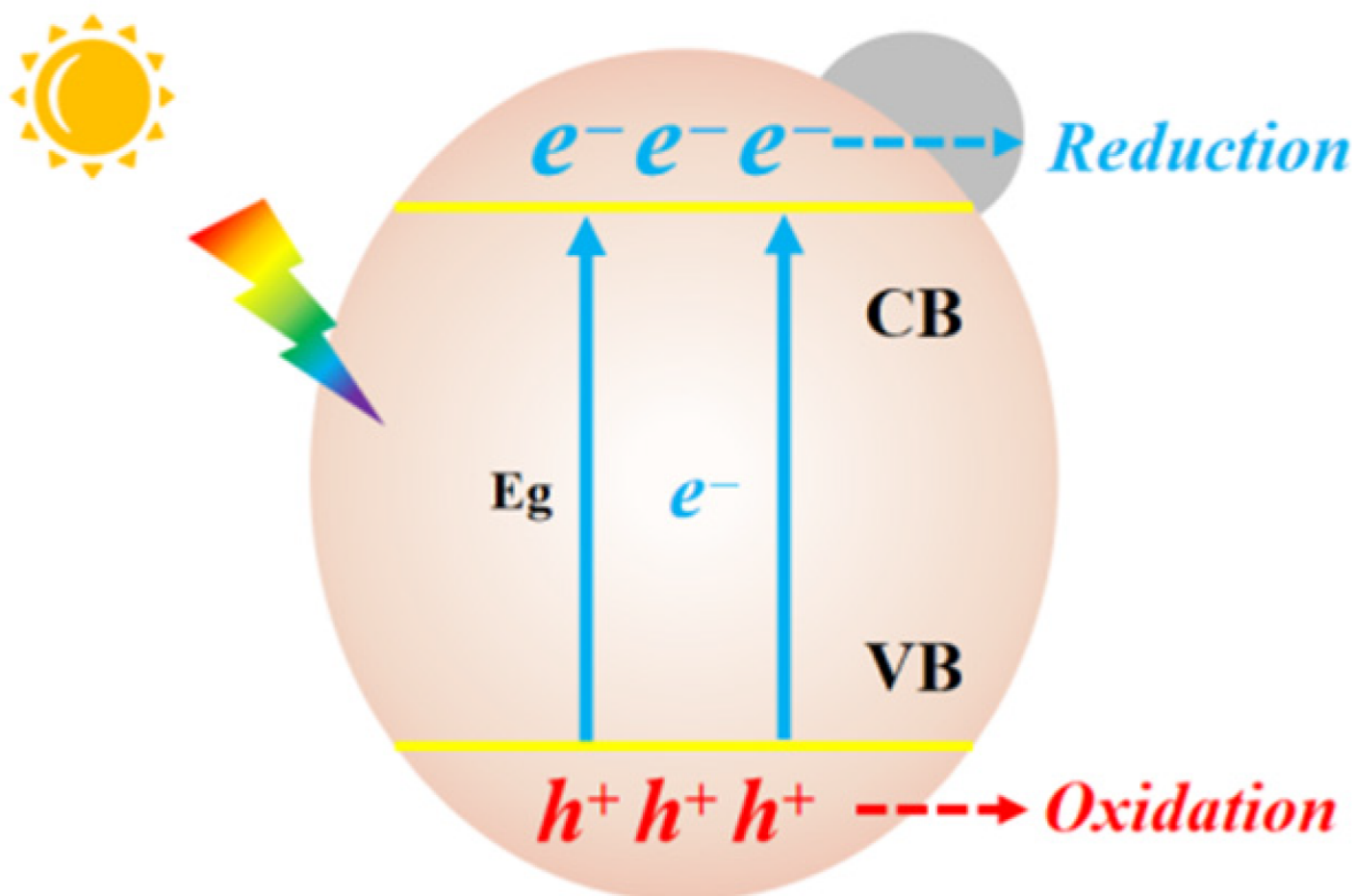
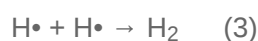
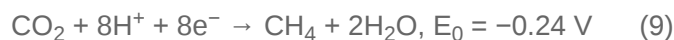


Figure 1. Fundamentals of photocatalytic reduction on a semiconductor catalyst.

The photocatalytic process of CO₂ conversion on semiconductors can be divided into five general steps: (1) Formation of electron–hole pairs under light radiation, (2) Separation and migration of the electrons and positive holes, (3) Adsorption and activation of CO₂, (4) Redox reactions between surface-adsorbed species and electron–hole pairs, and (5) Desorption of the product. The electrochemical reactions with standard oxidation–reduction potentials (at pH 7 vs. NHE) are as follows:





To carry out the conversion, it is necessary to comprehend the structure of CO₂. Due to the great symmetry and high bond energy of 750 kJ/mol, it is particularly difficult to break the C=O double bond. Therefore, the transmutation between CO₂ and bent radical anion of CO₂^{•-} on the surface of the catalyst is widely recognized as the first step to activate CO₂ for subsequent reaction. Additionally, photoreactions occur favorably only when the CB position of the catalyst presents a more negative potential than the target reduction and the VB position is more positive than the oxidation reaction. **Figure 2** summarizes the current strategies that can boost the photocatalytic CO₂ reduction pace, with detailed information being presented sequentially below.

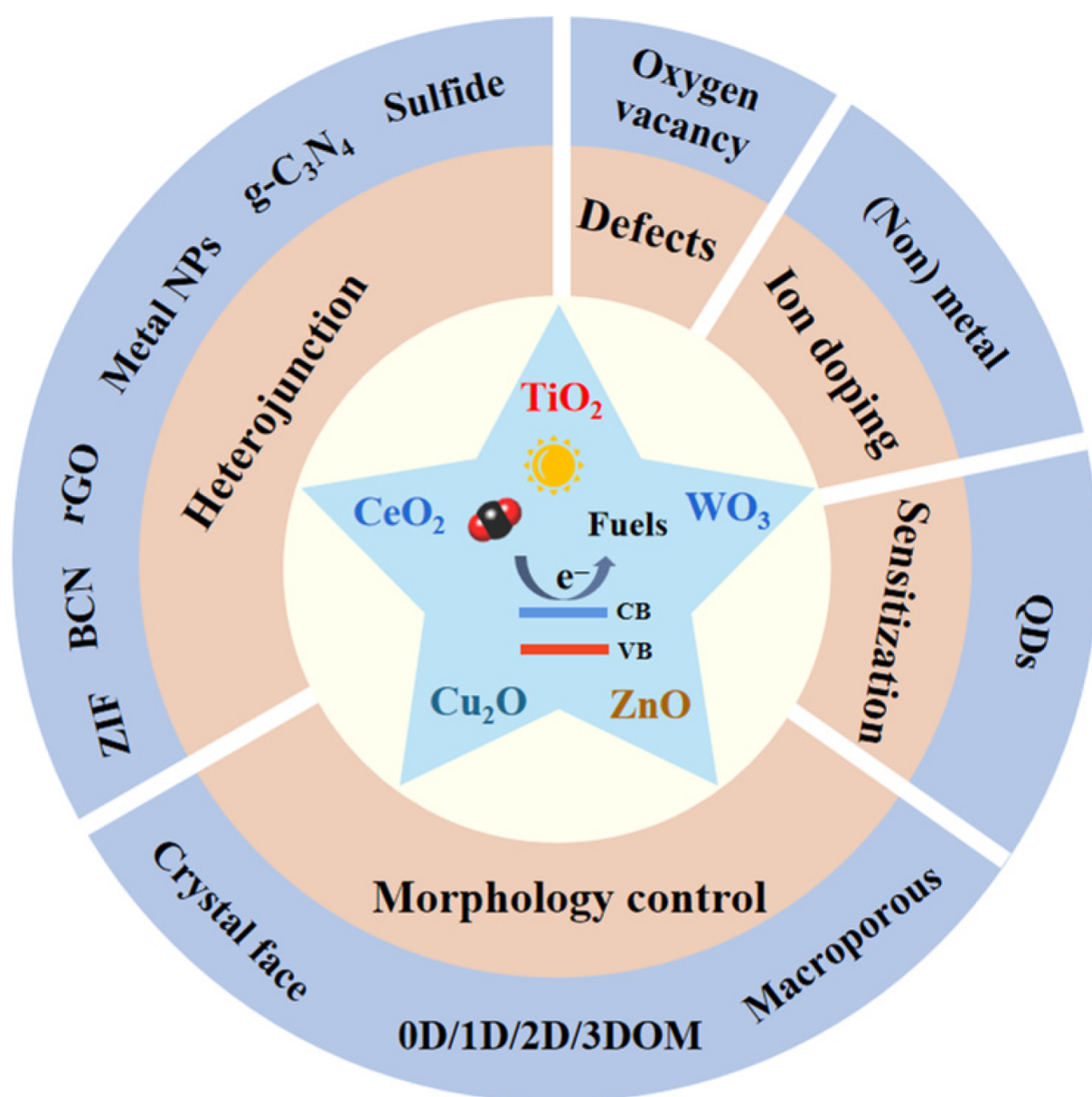


Figure 2. Strategies to boost the photocatalytic CO₂ reduction.

2.1. Morphological Control

The surface topography of nanocrystals could evidently alter the electronic structure, surface energy and chemical properties of catalysts. Therefore, morphological control is one of the most important issues to researchers in nanoscience, chemistry and physics. Open facets and edges determine the shape of nanocrystals. Thus, the preferential adsorption of additives on certain crystal surfaces provides a good opportunity to tune the surface of nanomaterials.

2.1.1. Exposed Facet Adjustment

Exploring and figuring out the variation that is connected with exposed surfaces is crucial to elucidate shape-related chemical and physical properties. In semiconductor crystals, different facets have distinct electronic band structures that influence the transport of photoexcited carrier charges. It is wise to expose active facets to tune the CO₂ photoreduction efficiency [6].

2.1.2. Quantum Dots (QDs)

Zero-dimensional (0D) semiconductor quantum dots (QDs) have many unique properties, such as quantum confinement effect, high extinction coefficient and multiple exciton generation [7]. Hence, QDs show much better photoactivity in the visible light region. Unlike bulk materials, surface atoms make up the majority of QD semiconductors. The abundant surface sites enhance the interaction between electron donors and acceptors, thus facilitating the photocatalytic charge transfer.

2.1.3. One-Dimensional and Two-Dimensional Structures

One-dimensional nanostructured catalysts have high aspect ratios, such as nanowires, nanorods and nanotubes. The morphological tuning of the material makes a difference to their thermal, optical, electrical and magnetic properties [8]. For instance, TiO₂ nanotube can act as a channel for electron transfer and build up the chemical reactions rate.

Two-dimensional layered materials can protect a tiny particle component from aggregation. The CO₂ adsorption capacity on 2D photocatalysts can be enhanced due to the large specific surface area and bountiful surface defects [9]. Two-dimensional lamellar nanosheets are widely used in photocatalysts, such as g-C₃N₄, MoS₂ and WO₃.

2.1.4. Macroporous and Three-Dimensional Ordered Macroporous (3DOM) Structures

Macroporous materials are widely used in photocatalytic materials, owing to their excellent properties [10]. Unlike dispersed particles, sunlight can penetrate the pore wall easily and scatter widely inside the hollow structure, thus increasing the efficiency of illumination. Subsequently, the slender walls of pore reduce the transfer length of photo-

generated charge carriers. Electrons (e^-) and holes (h^+) are separated more efficiently when heterojunctions are loaded on porous materials. The specific surface it provides is so large that more CO₂ molecules have a chance to contact the catalyst for reduction reactions.

Growing attention has been paid to hierarchical composite pores, including photonic crystal catalysis and separation of sub-microns. The slow light effect of photons associated with 3DOM materials have been considered to increase solar radiation absorption and enhance photocatalyst performance [11]. 3DOM products with periodic macrostructures [12], known as inverse opal, have been applied in battery materials, sensors, separation engineering and heterogeneous catalysis.

2.1.5. Preparation of 3DOM Materials

Specifically, 3DOM materials are produced by the colloidal crystal template (CCT) method (**Figure 3**). Firstly, a uniform and close-connected organic sphere template can be obtained, through three key processes. Then, seeping metallic salt sol into the void of microspheres. After heat treatment, the organic microsphere template fades away and leaves a metal oxide frame. (i) The polymerizable monomer (methyl methacrylate, styrene) and initiator are mixed and heated under the protection of Ar; (ii) the earlier reaction liquid is filtered with microfiltration membrane; (iii) the microsphere mixture is centrifuged at a high speed for a long time, yielding the polymethylmethacrylate (PMMA) or polystyrene (PS) template; (iv) the template is immersed in the precursor solution and (v) is calcinated.

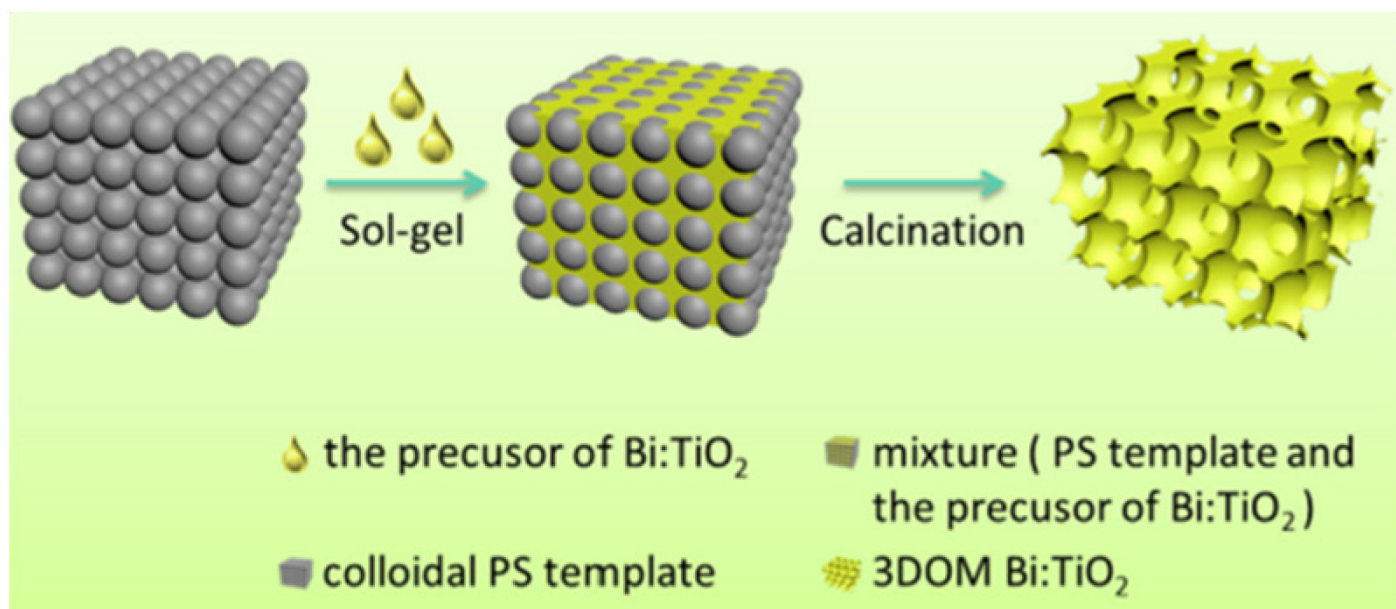


Figure 3. The fabrication process of the 3DOM Bi-doped TiO₂ [13].

2.2. Heterojunction

Photogenerated charge carriers in a single material tend to recombine rapidly due to Coulombic force and they can be separated efficiently in the multi-materials that are in close contact [14]. The heterojunction is an interface

between various semiconductors with different energy band structures. The heterojunction structure can reduce the probability of charge recombination and is therefore considered an effective method to enhance the photocatalytic activity [15][16].

2.3. Defects

All semiconductors have surface defects, which are rooted in the absence of host atoms. The amount of oxygen vacancies can be altered by ion doping and nanocrystal modifications. Oxygen defects and other defects [17] can facilitate the separation of h^+/e^- . Over and above, defects can also activate CO₂ molecules, reducing the activation energy of the reaction [18].

2.4. Ion Doping

Elemental doping is a common strategy to modulate the surface electronic structure. Then, the band gap of the semiconductors make a difference, nonmetal doping (such as N, C and O) mainly alters the VB and metal-doping (e.g., Mo, Co and Ni) influences the CB [19][20]. For example, Cu-doped TiO₂ absorbs more visible light due to the Cu 3d-Ti 3d optical transition [21].

2.5. Sensitization

Sensitization means coupling the quantum dots, dyes, etc., with semiconductors to increase the photogenerated carriers and promote the absorption of light by taking advantage of their receptivity in UV, visible or infrared light.

3. Photocatalysts with Different Basis Matrices

3.1. TiO₂-Based Photocatalysts

Titanium dioxide (TiO₂) is notable in photochemistry, with advantages such as non-toxic, cheapness, corrosion resistant, good physical and chemical stability. However, owing to the wide E_g of 3.0–3.2 eV, TiO₂ only absorbs energy in the ultraviolet region (3–5% of the solar energy), and photoexcited charge pairs are easy to combine, resulting in quantum inefficiency. Usually, TiO₂ is divided into the rutile, anatase and brookite on the basis of atomic arrangement modes. Rutile TiO₂ is thermodynamically stable and does not distort or decompose at high temperatures. It has a narrower energy gap (3.0 eV) and a wider spectral response than the anatase phase (3.2 eV). Rutile TiO₂ seeds generally grow larger in size and tend to form an agglomerated structure. Smaller anatase TiO₂ particles have a wider lattice gap and abundant surface oxygen defects, which make it favorable for ion doping and photoreactions.

3.1.1. Morphological Control

In hollow nanotube-shaped catalysts, the transport speed of CO₂ and photoproducts can be facilitated. Ru phase is inclined to form methane in CO₂ hydrogenation process [22] and Yang et al. [23] entrapped Ru nanoparticles in TiO₂

nanotubes. Restricted Ru nanoparticles were resistant to sintering and leaching in the Ru-in/TNT catalyst channel (Figure 4). Electrons tend to gather in the tubes because of a confinement effect, which leads to an abundant, accessible metallic phase. It is easier for high-priced Ru species to combine with free electrons and then exhibit a superb CH₄ and CH₃OH yield.

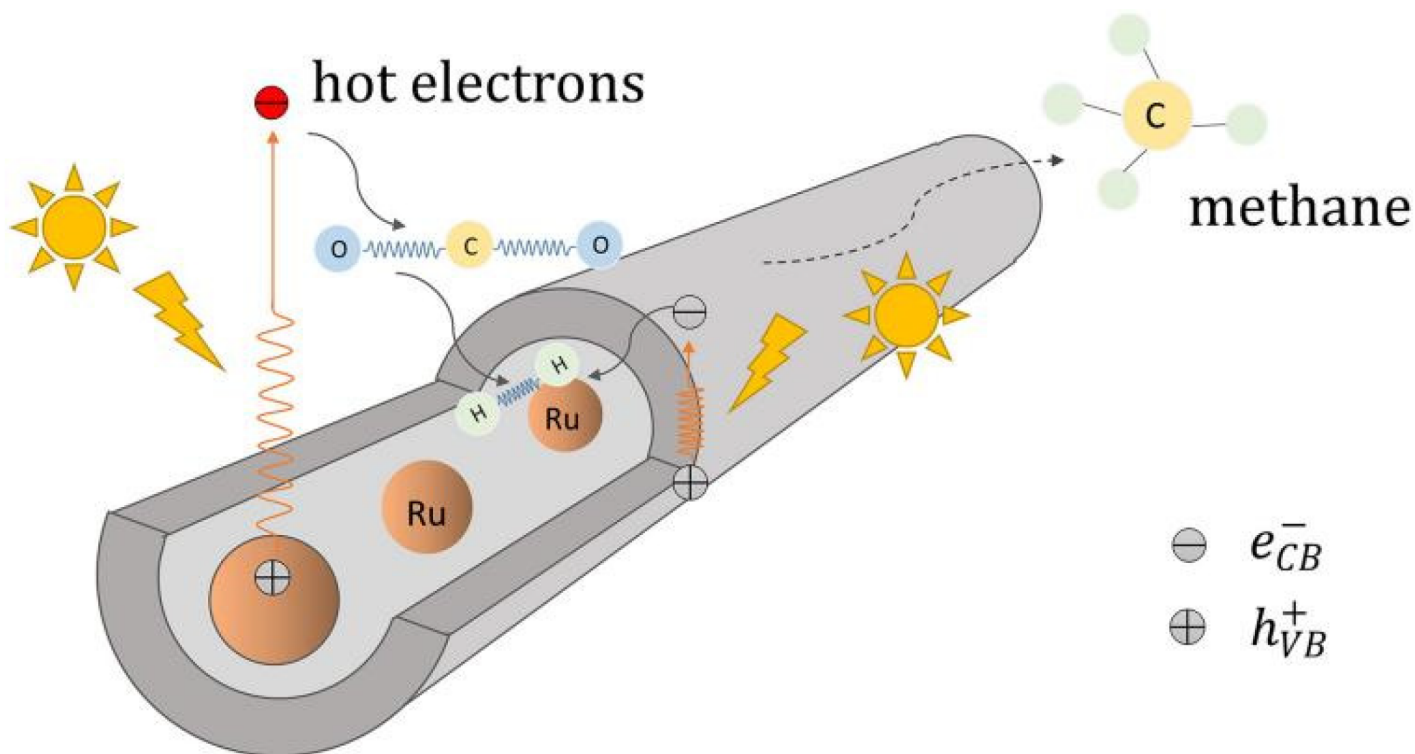


Figure 4. The reaction pathway for the Ru entrapped in the TiO₂ nanotubes catalyzes CO₂ methanation [23].

On the contrary, Kar et al. [24] loaded metal nanoparticles onto vertically oriented 1D TiO₂ nanotube arrays (TNAs) platforms via the graft method. In the synthesis process, Au, Ru and ZnPd NPs grow anodically on transparent glass substrates. There is no band bending phenomenon in Au NP-grafted TiO₂, which can be observed from ultraviolet photoelectron spectroscopy (UPS). TPD experiments proved that all NP-grafted samples absorb more CO₂ than TNAs. It is worth noting that, in nanoparticle-grafted TNAs, blue photons close to and below the TiO₂ band edge were excited to drive CO₂ photoreduction process. Ru-TNAs, ZnPd-TNAs and Au-TNAs had the CH₄ formation rates of 26, 27 and 58 μmol·g⁻¹·h⁻¹, respectively.

Metal–organic frameworks (MOFs), also known as porous coordination polymers, composed of organic linkers and metal nodes (metal ions or clusters), are a kind of porous crystalline inorganic–organic hybrid materials. The unique advantages of MOFs, such as extremely high surface area, uniform adjustable porous structure and high density coordination unsaturated metal sites, have been extensively researched.

Exposed Facet Adjustment

A Pt-TiO₂ single atomic site catalyst (PtSA/Def-s-TiO₂) was prepared [25] by the “thermal solvent-argon treatment and hydrogen reduction” method. In order to construct Ti–Pt–Ti structures, TiO₂ nanosheets with oxygen deficient sites were used to anchor monatomic Pt particles, which can retain the stability of isolated single atomic Pt and improve photocatalytic performance. The exposed (101) and (001) crystalline of TiO₂ nanosheets were determined by transmission electron microscopy, and a thickness of 6.9 nm was observed through atomic force microscope. The EPR spectra of the samples confirm that the rich oxygen defect structure can be obtained by heating TiO₂ nanosheets in argon atmosphere.

3DOM Structure Ti-Based Materials

A ternary 3DOM Bi-doped TiO₂ photocatalyst decorated with carbon dots (CDs) was obtained, whose pore engineering of the 3DOM skeleton greatly promoted the response in the whole solar spectrum range [13]. It exhibits enhanced photocatalytic performance because of its excellent exquisite structure and high charge transfer efficiency. Similarly, a BiVO₄/3DOM TiO₂ nanocomposite [26] was synthesized as a highly efficient photocatalytic catalyst for the degradation of dye pollutants.

3.1.2. Heterojunction

p–n heterojunction: A p–n heterojunction is formed by combining p-type and n-type semiconductors. Even without light irradiation, electrons can diffuse from an n-type semiconductor to a nearby p-type one, in the case of the combination of two materials. Correspondingly, the holes on the surface of a p-type semiconductor are transferred to the n-type one, which results in an efficient separation of charge carriers. A ZnFe₂O₄-modified TiO₂ was synthesized by the hydrothermal method [27], and the p-n heterojunction system could reduce CO₂ to methanol at a yield of 75.34 μmol g⁻¹ h⁻¹.

rGO composite: graphene materials have been widely used because of their large specific surface area, unique thermal stability and excellent electrical conductivity. Graphite nanomaterials are visible-light-responsive materials with appropriate band gaps, and the energy levels of CB and VB are in optimal positions relative to ordinary hydrogen electrodes. These unique photocatalytic properties have made them prime candidates for photocatalytic CO₂ reduction. Fortunately, tightly contacted ultra-thin graphene layers and TiO₂ compounds and can be prepared with some additives [28].

3.1.3. Ion Doping

In recent years, elements such as B, N, Co and Bi have been widely applied in TiO₂ doping. A carbon-based hybrid nanocomposite reduced graphene oxide (rGO), belonging to the narrow band gap, with oxygen-containing functional groups on the surface that can be enhanced by π interactions [29]. Laminate graphene carriers not only prevent TiO₂ repolymerization, but also hybridize the function of the catalytic system. Co-doped TiO₂ was loaded on the rGO [30], and the Co peak in EDX spectra and C-O peak in FT-IR spectra confirmed the successful doping and the presence of graphene support, respectively. The size of TiO₂ particles decreased from 48–80 nm to 23–28 nm, which is consistent with earlier reports of changes in titanium doping with transition metal ions.

3.1.4. Sensitization

A growing number of semiconductor materials are used to modify TiO₂, but randomly mixed catalysts are not stable enough to achieve reproducibility. Therefore, Lee [31] grew well dispersed p-type NiS nanoparticles on the surface of a highly aligned n-type TiO₂ film to obtain the NiS-sensitized TiO₂ films. The band gaps of two components were estimated by wavelength relation. Some inferences can be drawn when considering the results of both the ultraviolet and visible spectra. It indicates that more electrons are subpoenaed from the short- E_g NiS and transferred to TiO₂ conduction band. The spectra results reconfirmed the electron contribution of the NiS and the design of a catalyst that produced 3.77-fold CH₄ compared to the TiO₂ film.

3.2. WO₃-Based Photocatalysts

Tungsten-based oxides (WO₃) have been extensively studied in recent decades and various morphologies have been presented. In the WO₃ structure, the crystal in the stoichiometric ratio is connected with a twisted WO₆ octahedra to form a perovskite crystal structure. It has monoclinic, orthorhombic and hexagonal crystal forms. At the same time, the oxygen lattice can be lost easily, resulting in oxygen vacancies and coordinatively unsaturated W atoms. Therefore, tungsten oxide has many non-stoichiometric compounds, such as WO_{2.72}, WO_{2.8}, WO_{2.83} and WO_{2.9}.

3.2.1 Morphological Control

Bi₂WO₆ is one of the tungsten-based materials that belongs to Aurivillius crystal oxides. Its crystal has an orthorhombic system, and its narrow band gap (2.7–2.9 eV) structure allows it to meet the response absorption of visible light. Moreover, its stable structure and eco-friendly properties have attracted many scientific researchers to study it. Since the valence band of Bi₂WO₆ is composed of O_{2p} and Bi_{6p}, and the conduction band is composed of W_{5d}-assisted Bi_{6p} orbitals, the VB energy levels can be dispersed broadly. By employing the Kirkendall effect in ion exchange and BiOBr precursor, Huang et al. [32] prepared a bowl-shaped Bi₂WO₆ HMS material. Based on the large specific surface area of the material, its adsorption capacity for CO₂ reaches 12.7 mg g⁻¹ at room temperature and pressure. The material adsorbs a large number of HCO₃⁻ and CO₃²⁻ species on the surface during the reaction, which makes the catalytic reaction easier. The Bi₂WO₆ HMS thus has a high catalytic activity, and the methanol yield is 25 times higher than that of the Bi₂WO₆ SSR.

Iron phthalocyanine FePc is neatly assembled on porous WO₃ under induction and coupled with surface atoms by H-bonding [33]. The optimized FePc/porous WO₃ nanocomposites exhibit enhanced CO₂ photoreduction activity, which is attributed to the synergistic effects of a high specific surface area, a better charge separation and proper central metal cation. A series of mesoporous WO₃ with interconnected networks were synthesized by the silica KIT-6 hard template method [34], which became oxygen-deficient after hydrogenation treatment. Both the ordered porous structure and oxygen vacancies contributed to the increased yield of CH₄ and CH₃OH.

WO₃ with a hollow nest morphology with hierarchical micro/nanostructures (HNWMs) was synthesized [35] by the one-step hydrothermal method (**Figure 5**), with a particle diameter of about 2.5 μm. The 2D nanosheets, which

have an average thickness of 30–40 nm, were assembled to build a distinctive hollow nest structure with a good stability and reusability under visible light. Hao et al. [36] prepared core–shell heterojunctions of two-dimensional lamellar WO₃/CuWO₄ by the in situ method. After the modification of amorphous Co-Pi co-catalyst, the photoanode of ternary homogeneous core–shell structure exhibited a high photocurrent of 1.4 mA/cm² at 1.23 V/RHE, which was 6.67 and 1.75 times higher than that of the pristine WO₃ and 2D homogeneous heterojunction.

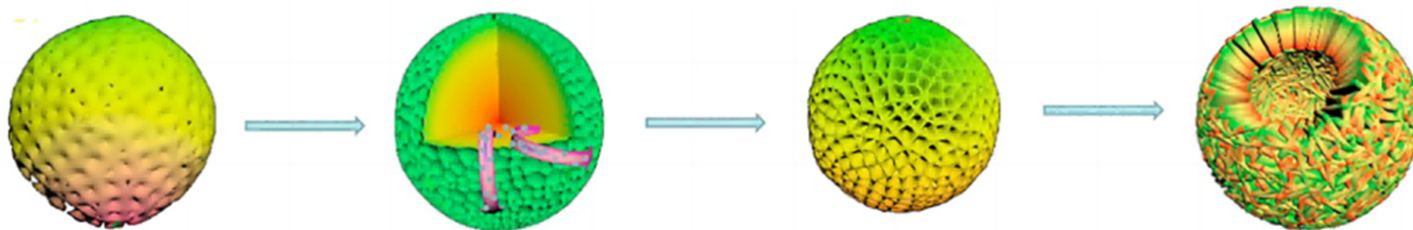


Figure 5. Schematic diagram of the HNWM formation process [35].

Ti atoms in ultrathin Ti-doped WO₃ nanosheets promoted the charge transfer [37], as they accelerate the generation of key intermediates COOH*, which was revealed by in situ characterization. Furthermore, Gibbs free energy calculations were calculated to verify that ion doping can reduce the CO₂ activation energy barrier and CH₃OH desorption energy barrier by 0.22 eV and 0.42 eV, respectively, thus promoting the formation of CH₃OH. The ultrathin Ti-WO₃ nanosheets showed an excellent CH₃OH yield of 16.8 μmol g⁻¹ h⁻¹. Two-dimensional bilayered WO₃@CoWO₄ were prepared [38] via a facile interface-induced synthesis method. The optical energy conversion efficiency can be improved by both p–n heterojunctions and interfacial oxygen vacancies. The narrow band gap of the WO₃@CoWO₄ heterojunction was proved by DFT calculations and some characterizations, which allows a better visible light absorption. A tree-like WO₃ film was prepared [39] by the hydrothermal process, which has a large specific surface area. The WO₃ product was a unity of hexagonal/monoclinic crystals, which contained W⁵⁺ defects and oxygen vacancies. The products were further subjected to a mild reduction solution at the lower temperature of 333 K to introduce more defects.

Preferentially Exposed Facets

According to studies, infrared (IR) light makes up nearly 50% of solar energy, and it is challenging to make use of the majority of the light. Liang et al. [40] fabricated 2D ultrathin WO₃ with an intermediate band gap. They achieved the first complete decomposition of CO₂ driven by infrared light without the addition of sacrificial agents. Theoretical calculations indicated that the generation of the intermediate energy band resulted from the critical density of the generated oxygen vacancies, which has also been verified by synchrotron valence band spectroscopy, photoluminescence spectroscopy, ultraviolet-visible-near-infrared spectroscopy and synchrotron infrared reflection spectroscopy.

3DOM Structure W-Based Materials

Unexpectedly, it was found that the resistance of 3DOM-WO₃(270) and the Ag₃PO₄ electron absorption band were comparable. By depositing Ag₃PO₄ nanoparticles in the micropores of 3DOM-WO₃, Chang et al. [41] achieved a

higher photocatalytic activity and more efficient light harvesting at the wavelengths of 460–550 nm. A Z-scheme g-C₃N₄/3DOM-WO₃ catalyst designed by Tang et al. [42] also has a high CO₂ photoreduction activity.

3.2.2. Heterojunction

Quantum dot composite: CuO quantum dots (QDs) were combined with WO₃ nanosheets by a self-assembly method and the diameter of 6%CuO QDs/WO₃ NSs was mainly located at 1.6 nm [43]. The bandgap energy of CuO/WO₃ fell in 2.28 eV and the complex catalyst possessed a lower resistance for charge carrier transfer that showed in UV-vis DRS and EIS analysis. Due to the low CB position, CO cannot be obtained when using pure WO₃. However, the photogenerated electrons gathered in the WO₃ CB position was able to reach the CuO VB position when the Z-scheme (Figure 6) was formed by intimate heterojunctions. At the same time, the reduction reaction that transformed CO₂ into CO occurred at the CuO CB position.

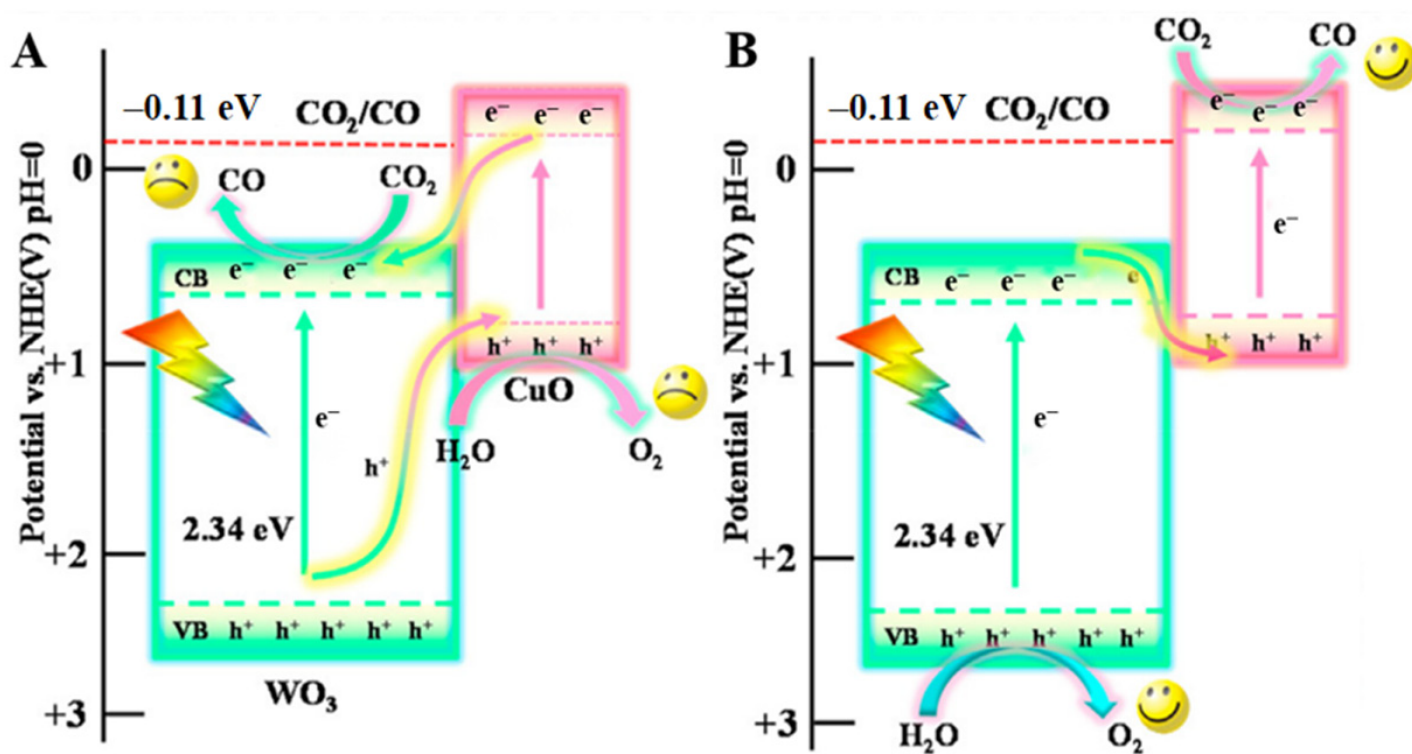


Figure 6. The proposed charge transfer mechanisms: (A) II-scheme and (B) Z-scheme for CuO/WO₃ [43].

3.2.3. Ion Doping

Molybdenum with a similar ionic size was chosen to dope the WO₃ as a low-valence metal species. The W⁵⁺/W⁶⁺ ratio of the catalyst was increased, making it easier to exchange electrons with reactants. The conductivity of protons was enhanced by the presence of hydrogen bronze, which originated from a chemical reaction between WO₃ and Brønsted protons and excess electrons in their lattices.

3.3. ZnO-Based Photocatalysts

ZnO, a common metal oxide, is a n-type semiconductor with an E_g value of 3.37 eV. It is a kind of amphoteric oxide that has the advantages of nontoxic harmless, low cost, abundant reserves, convenient preparation, low dielectric constant and low optical coupling rate. ZnO has three main lattice structures: wurtzite structure, zinc-blended structure and tetragonal rock salt structure. The wurtzite structure is considered the most stable and common structure in nature. It is a kind of hexagonal crystal, in which the O and Zn atoms are aligned with the hexagonal density stacking.

3.3.1. Morphological Control

The 3nm Pt particles were uniformly dispersed over ZnS@ZnO with a mesoporous heterostructure [44] and more CH₃OH was obtained. Reactant charge carriers entered the pore channels of the porous heterozygous layer, thus reducing the likelihood of flow resistance and electron–hole recombination. The S-scheme photocatalyst delivered a high CH₃OH formation rate of 81.1 $\mu\text{mol g}^{-1} \text{h}^{-1}$, which is roughly 40 and 20 times larger than that of bare ZnO (3.72 $\mu\text{mol g}^{-1} \text{h}^{-1}$) and ZnO–ZnS (4.15 $\mu\text{mol g}^{-1} \text{h}^{-1}$). On the other hand, a porous ZnO@ZnSe core/shell nanosheet array material (Figure 7A) was prepared in a controlled manner [45]. The final n-type semiconductor composites had a proper negative CB band edge. In comparison to ZnO or ZnSe, more pairs of electron–holes were formed under visible light. Electrons tend to land on ZnO, which is aimed at methanol production. Mei et al. prepared a ZnO microsphere with different numbers of shells [46] and the photoelectric performance of ZnO was optimal when the number of shells reached three.

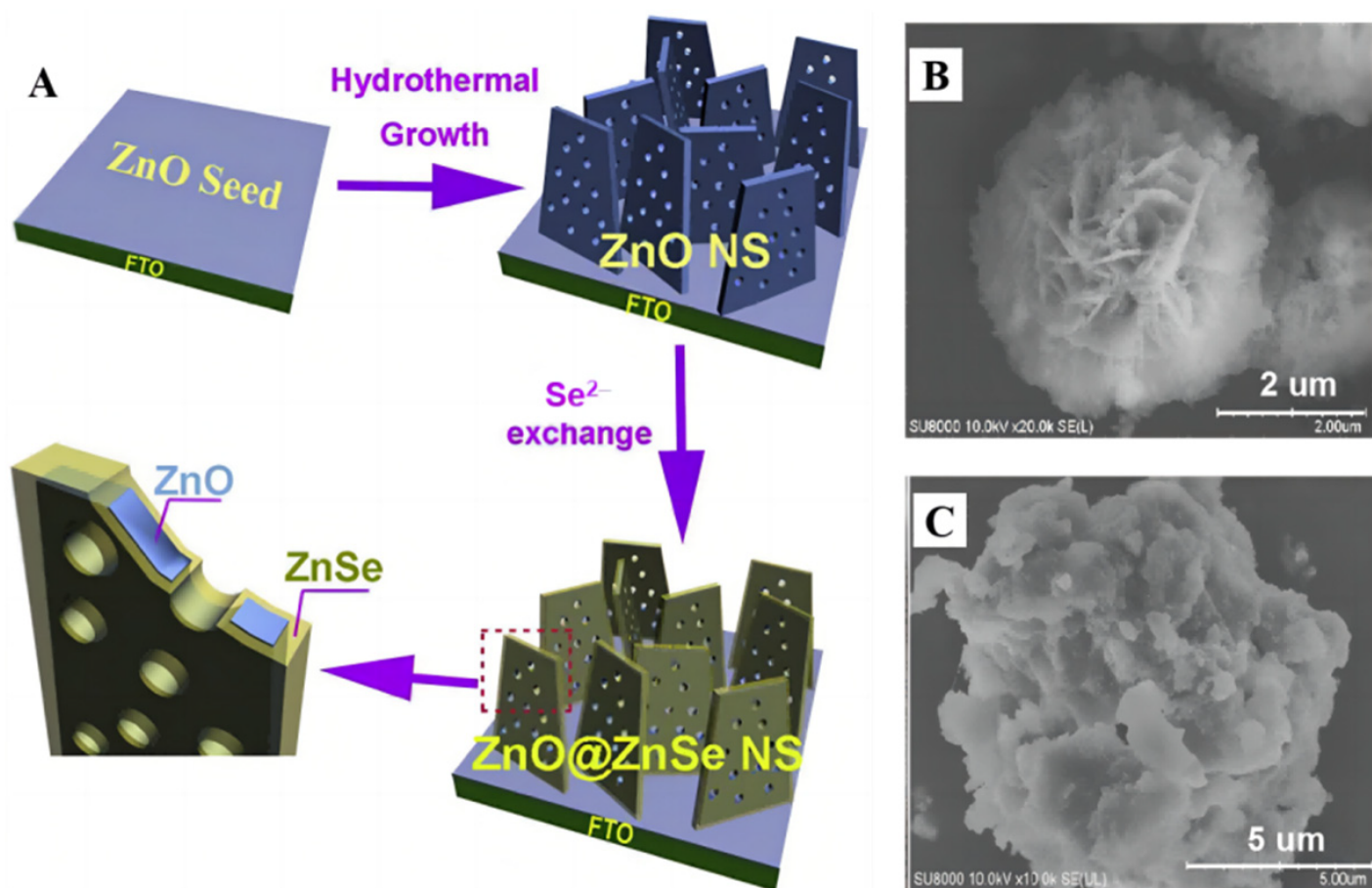


Figure 7. (A) Schematic illustration of the fabrication process of ZnO@ZnSe photocathode [47]. (B) ZnO/ZnS (C) ZnO/ZnS/g-C₃N₄ SEM images [45].

ZnO/ZnS nanoflowers (Figure 7B) were combined with g-C₃N₄ nanosheets (Figure 7C) to construct a double Z-scheme structure [48]. ZnO/ZnS nanoflowers provide a large specific surface area and g-C₃N₄ helps to absorb more photons under solar light irradiation. Optimized interfacial charge transfer dynamics in ternary heterostructure can be characterized by photocurrent measurements. As a result, the formation rate of H₂ product over the novel double Z-scheme mixture increases to 301 μmol g⁻¹ h⁻¹ on water splitting.

3DOM Structure Zn-Based Materials

Wang et al. [49] published metal-organic-framework-derived 3DOM N-C doped ZnO (Figure 8) for efficient CO₂ reduction. The ultra-tiny CoO_x clusters were anchored on the surface of catalyst and no Co-Co peak was found in CoO_x/N-C-ZnO. The charge transfer rate was jacked up by ion doping and the recombination of electron-hole pairs was tamed because of the CoO_x clusters. Furthermore, CoO_x on the orderly connected channels can act as an electron trap to capture electrons, which makes a contribution to photoreaction efficiency. The density theory calculations (DFT) was also used to detect the CO₂ adsorption ability, and CoO_x/N-C-ZnO exhibited the most negative CO₂ binding energy due to improved electron structure of adsorption site.

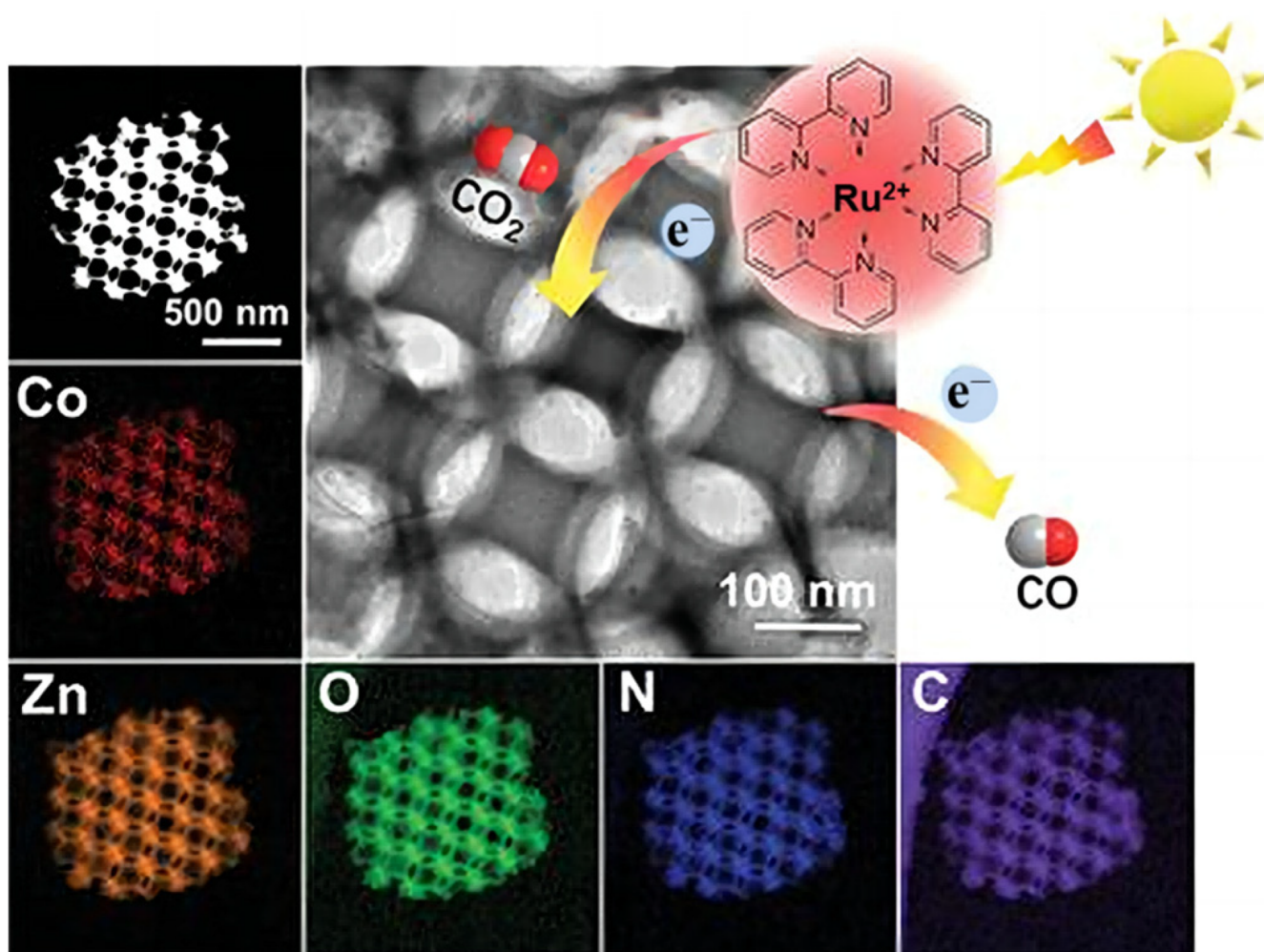


Figure 8. HAADF-STEM image of CoO_x/N-C-ZnO and related elemental mapping images [49].

3.3.2. Heterojunction

Recently, zeolitic framework (ZF) composite fabricated by the microwave-hydrothermal synthesis method (MWH) has attracted attention, which can provide a fast heating-speed and produce morphologically uniform samples. With biodegradable template, the zeolitic framework (ZF) was synthesized via MWH method from volcanic ashes. The NaAlSiO₄ (NAS) framework was composed of 50 nm circular channels and has a large surface area. The synergistic effect among ZnO, CuO and ZF support accelerated the photocatalytic process of CO₂ reduction, which offers higher HCOOH evolution rate.

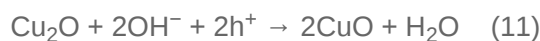
3.4. Cu₂O-Based Photocatalysts

Cuprous oxide (Cu₂O) is a potential p-type semiconductor with a wide visible-light response range and high photo-electric conversion efficiency (18%) [50], and it displays attractive prospects in solar energy conversion and heterogeneous photocatalysis. Although Cu₂O possesses many excellent properties, photocorrosion and the rapid recombination of e⁻/h⁺ pairs affect its activity and limit its application. The photocorrosion is believed to occur in two ways: (1) self-reduction caused by generated electrons and (2) self-oxidation caused by the generated holes.

Self-reducing photocorrosion:



Self-oxidative photocorrosion:



Therefore, developing Cu-based catalysts with excellent activity, selectivity and stability has become the research hotspot in the area of the photocatalytic reduction of CO₂. Many successful attempts have been made to improve the photostability and photocatalytic performance of Cu₂O. In general, most studies focus on enhancing the charge transfer from Cu₂O to reactants or cocatalysts to prevent charges from accumulating within the particles.

3.4.1. Morphological Control

A branch-like Cd_xZn_{1-x}Se nanostructure was obtained [51] by the cation-exchange method, which was then mixed with Cu₂O@Cu to form heterojunctions. Selenium (Se) vacancies were created during the ion exchange process and the crystal growth was limited due to the additive diethylenetriamine (DETA), leading to insufficient coordination of the surface atoms, which then become active adsorption sites. Highly hierarchical branching-like structures assembled by one-dimensional structural materials not only facilitate electron accumulation at their tips but also increase the light-accepting area, and characterization results show that branching structures can effectively absorb visible light. Cd_{0.7}Zn_{0.3}Se/Cu₂O@Cu step-scheme heterojunction exhibited a CO release yield of 50.5 μmol g⁻¹ h⁻¹.

Ultrafine cuprous oxide U-Cu₂O (<3 nm) was grown on the polymeric carbon nitride (PCN) (**Figure 9**) by the in situ method [52]. PCN has a narrow band gap of 2.7 eV that can capture visible light. Both ultrafine nanoclusters and Z-scheme heterojunction can protect U-Cu₂O from degradation. The photocatalyst U-Cu₂O-LTH@PCN has high stability, maintaining more than 95% activity after five cycles of testing, while bare Cu₂O grades completely within three cycles. A large number of heterojunctions were formed by U-Cu₂O particles and lamellar PCN, expediting the electron transfer efficiency. The product can convert CO₂ to methanol with water vapor under light irradiation at the high yield of 73.46 μmol g⁻¹ h⁻¹.

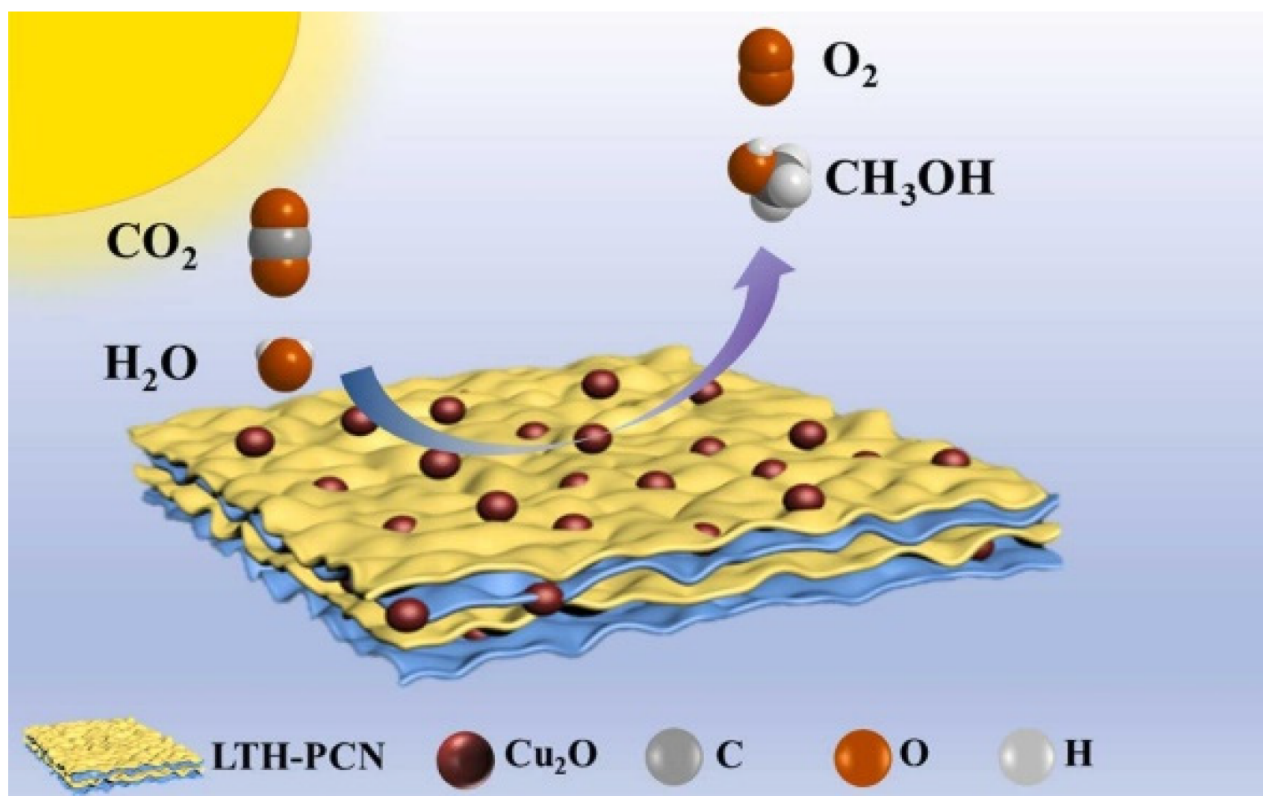


Figure 9. Ultrafine U-Cu₂O nanoclusters anchored on the photosensitizing PCN support [52].

Preferentially Exposed Facets

Cu₂O is an ideal compound to study the influence of electron-related effects. The rare occurrence of the O-Cu-O 180° linear coordination of Cu₂O makes its (111), (100) and (110) facets chemically active. Zhang et al. [53] successfully achieved the morphology control of Cu₂O nanocrystals by utilizing the selective surface stabilization of PVP on the (111) plane of Cu₂O. With different amounts of PVP, the surface area ratio of (111) to (100) was subtly tuned, which resulted in the shape evolution of the system and various Cu₂O structures (**Figure 10**). The detailed modification mechanism was elucidated from the structural and kinetic perspectives.

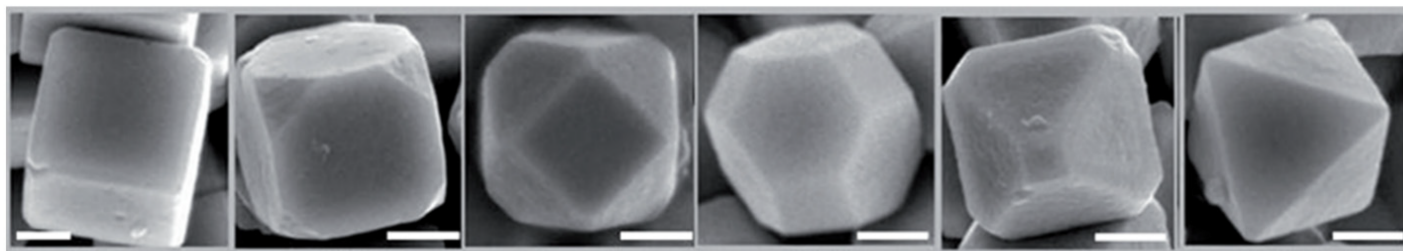


Figure 10. FESEM images of the Cu₂O polyhedrons with different volume ratios of (100) to (111) [53].

3DOM Structure Cu-Based Materials

The 3DOM Cu₂O structure was luckily obtained [54] via polystyrene crystal templates. Under the contrived “sunlight” irradiation, incident light was reflected and absorbed around and around again. In the ultra-visible absorption spectra (350 to 800 nm), Cu₂O with large orifices absorbs more photons than bulk samples, making it more advisable for solar applications. 3DOM Cu₂O was prepared [55] by the electrochemical method to reduce CO₂, and its Faraday efficiency was five times higher than that of Cu film. The CO₂^{•-} intermediate in 3DOM channels is more stable and leads to the possibility of forming CO and HCOOH products (Figure 11).

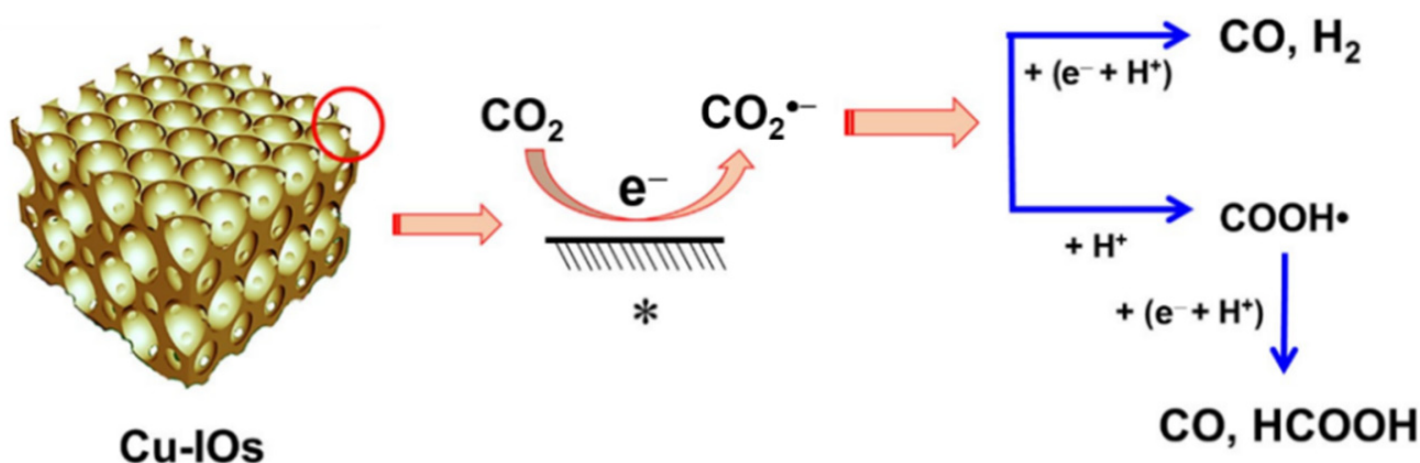


Figure 11. Proposed mechanism for CO₂ reduction to CO and HCOOH on Cu₂O-derived Cu-IOs (The symbol “*” represents the surface.) [55].

3.4.2. Heterojunction

Liu et al. [56] reported a facile solution and chemistry route to synthesize rGO-incorporated crystal Cu₂O with various facets as visible-light-active photocatalysts for CO₂ reduction. The enhanced activity was attributed to the formation of the heterojunction and the existence of rGO as the electron transport mediator. M. Flores et al. [57] adopted the microwave-hydrothermal method to couple the powders of Mg(OH)₂, CuO and Cu₂O. The synthesis method allowed a sufficient interaction between Mg(OH)₂/CuO and Cu₂O without inhibiting the gas adsorption capacity of Mg(OH)₂. They found that the presence of Cu₂O favored the selectivity towards CH₃OH production because a higher Cu⁺ concentration led to better selectivity.

3.5. CeO₂-Based Photocatalysts

Cerium oxide (CeO₂) has an octahedral face-centered cubic fluorite structure, in which the coordination numbers of Ce and O are 8 and 4. When reduced at a high temperature, it can be converted to nonstoichiometric CeO_{2-x} (0 < x < 0.5). Notably, CeO_{2-x} maintains a fluorite crystal structure and forms oxygen vacancies after losing a certain amount of oxygen. CeO_{2-x} materials with different Ce/O ratios were also obtained in different conditions and it could be reconverted to CeO₂ again if it returned to an oxidizing environment. Because of the unique electrical structure, cerium oxide (CeO₂) is famous for the conversion rates between Ce⁴⁺ and Ce³⁺, which have been studied as oxygen storage catalytic materials and solid oxide fuel cells by many scholars [58][59]. In summary, CeO₂ is a rare-earth metal oxide with a good photochemical stability, low cost and environment friendly characteristics.

3.5.1. Morphology Control

Yb-, Er-doped CeO₂ hollow nanotubes were synthesized [60] using silver nanowires coated with silica, and the products had a narrower band gap of 2.8 eV. The core-shell structured CeO₂ was converted into mesoporous hollow spheres by the Ostwald ripening method in the presence of urea and hydrogen peroxide [61]. CeO₂ nanocages can be fabricated by mixing (NH₄)₂Ce(NO₃)₂ with templates of Cu₂O nanocubes [62], in which Cu₂O is finally sacrificed. The photocatalytic results [63] indicated that CeO₂ nanocages exhibit higher activity than hollow spheres.

Preferentially Exposed Facets

It was found that molecular CO₂ can be distorted and participate in reactions at a low energy on the CeO₂ surface [64]. A p-type NiO material was designed to modify the rod-like CeO₂ nanostructure [65], allowing electrons and holes to migrate to opposite directions. They then operated the Mott-Schottky test, which showed a typical p-n junction. The presence of hexagon-shaped NiO plates broadened the range of light responses, which can be verified in the UV-Vis absorption spectra. Graphene oxide (rGO) was introduced as a “network” of for photoreduction electron transportation (**Figure 12A–C**). The impedance can be seen in the EIS Nyquist plot, which shows that the NiO/CeO₂/rGO achieved the minimum value. The HCHO production rate of the ultimate catalyst was 421 μmol g⁻¹h⁻¹ with the synergy of several favorable factors. It is worth mentioning that a range of in situ techniques have been used to detect oxygen vacancies, structural changes, free radicals and formate on the surface of CeO₂.

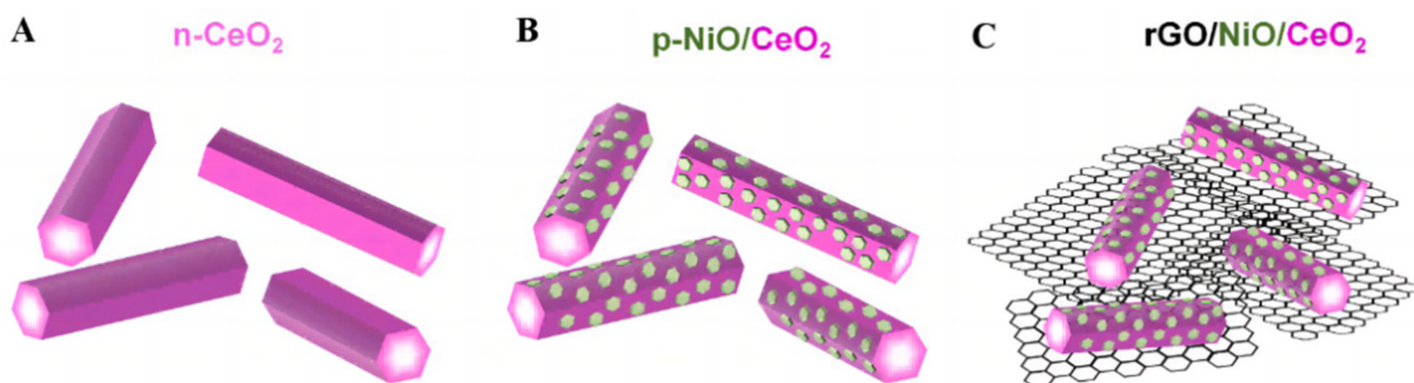


Figure 12. The structural diagrams of (A) n-CeO₂ nanorods, (B) p-NiO/CeO₂ composite and (C) NiO/CeO₂/rGO hybrid composite [65].

Macroporous

Mesoporous N-doped CeO₂(NMCe), a relatively ordered intermediate structure with enhanced CO₂-capturing capability, was prepared without any convoluted procedures or expensive equipment [58]. In the Roman spectrum, the bands from 550 to 650 cm⁻¹, which are closely related to oxygen vacancy, were more salient than the MCe band. In addition, N-doped porous CeO₂ has a higher CO₂ absorption capacity than porous CeO₂.

3DOM Structure Ce-Based Materials

Under the protection of poly alcohol, Zhang et al. [66] synthesized 3DOM CeO₂ that was loaded with Au–Pd alloys. 3DOM CeO₂ photocatalytic materials are expected to emerge in the field of CO₂ emission reduction, which could open up more possibilities for the development of super-catalysts.

3.5.2. Heterojunction

Researchers tried to combine CeO₂ with g-C₃N₄, which is popular for its energy bands and chemical stability. Through hard work, a three-dimensional porous g-C₃N₄(3DCN) was achieved, with the advantages of multi-channel structure. To accommodate more electrons and heighten the density of photoelectric currents, Zhao et al. [59] loaded Pt nanoparticles (5–6 nm) on CeO₂/3DCN using photodeposition techniques that require UV lamp radiation. The photoreduction rate gradually increased as the CeO₂ amount rose in the range of 15–45% and the yield rates of 4.69 and 3.03 μmol·h⁻¹·g⁻¹ for CO and CH₄ were achieved, respectively, after decorating with Pt crystalline grains.

4. Other 3DOM Materials

In order to introduce advanced porous structures to slow the self-aggregation of quantum dots, Wang et al. [67] devised 3DOM N-doped carbon (NC) to support CdS and ZnO QDs (**Figure 13**). They filled the interspace in an ordered PS microsphere template and then employed a pyrolytic treatment and in situ growth methods. Compared to bulky CdS, the 3DOM compounds have a larger cathodic current density and enhanced light harvesting, bearing a satisfactory carbon monoxide yield of 5210 μmol g⁻¹ h⁻¹. A three-dimensional SnO₂ inverse opal structure was synthesized as gas sensors, soot oxidation catalyst and photoanode. The 3DOM BiVO₄/SnO₂ heterostructure was obtained by adding a BiVO₄ precursor to fill the space between SnO₂ skeleton and periodic PS template. The compatibility of energy states with SnO₂ significantly reduced their photoluminescence intensity. Meanwhile, Au nanoparticles enhanced the slow photon effect, which in turn increased the incident light utilization efficiency.

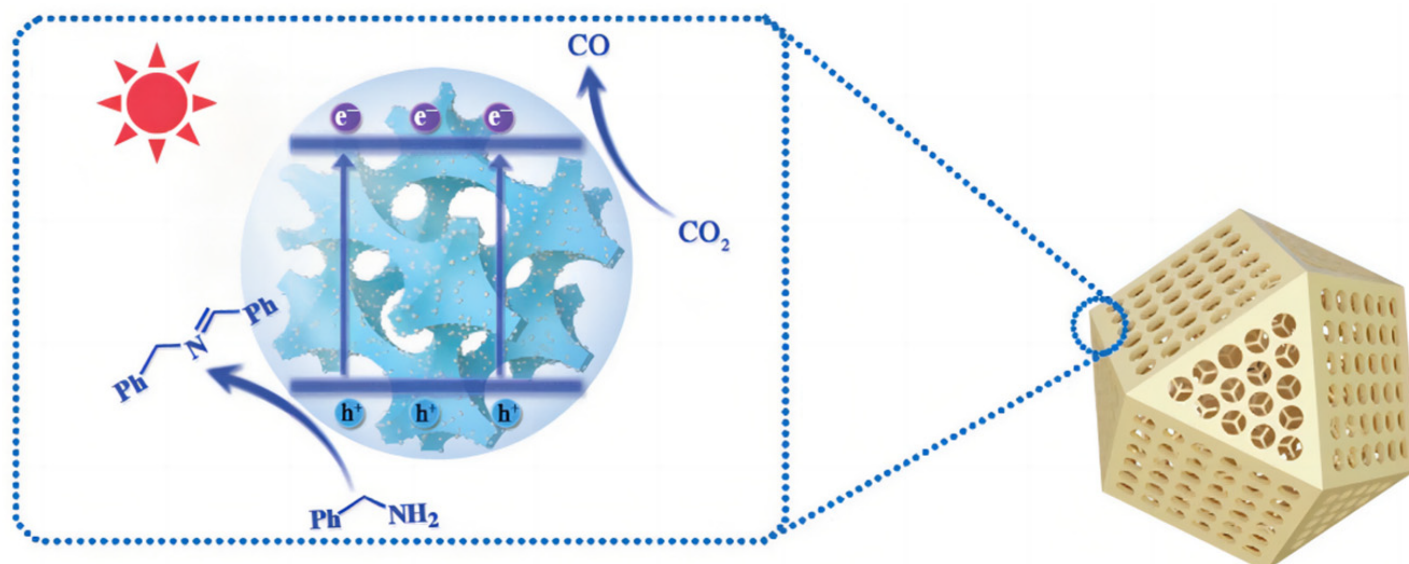


Figure 13. Proposed mechanism for the photocatalytic CO₂ reduction on 3DOM CdS QD/NC. Schematic illustration of the photocatalytic CO₂ reduction coupled with selective arylamine oxidation reaction system [67].

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