Zirconium and Titanium Phosphates

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Zirconium and titanium phosphates are a group of laminar tetravalent metal phosphates that have attracted intensive investigation interests. Unlike zirconium phosphates, titanium phosphates can be also synthesized in another morphology, exhibiting the form of nanorods, and thus are known as nanofibrous titanium phosphates. The different synthetic routes of zirconium and titanium phosphates and their current and prospective applications with a special mention to the their plausible applications in the field of biotechnology are discussed.

Keywords: zirconium phosphate; titanium phosphate; nanolayered phosphates; nanofibrous titanium phosphate; chemical synthesis; crystal structure; technological applications; drug delivery; biomedical applications

1. Zirconium Phosphates

Since the 1950s, many studies have focused on lamellar solids, particularly their properties and prospective applications. Special attention has been given to the development of innovative synthetic approaches for their preparation, and subsequent implementation in applications for different technological fields [1]. In this context, a group of laminar tetravalent metal phosphates is worth mentioning, with zirconium phosphates as remarkable structures that have attracted intensive investigation interests $\frac{[2][3][4]}{2}$. The great appeal of zirconium phosphates (hereafter abbreviated as ZrP) is mainly attributed to their robustness, the versatility of their crystalline structure, and the tunability of their crystallinity degree. In addition, the textural properties of ZrP particles can be controlled by modifying the synthetic conditions, or chemistry of their layers, generally by introducing functional groups on their surface. In fact, the scientific interest in ZrP goes back to the beginning of the 20th century, when the first trials of using soluble zirconium salts, e.g., $ZrOCl_2 \cdot 8H_2O$, in the elimination of phosphate ions in solution were reported $\frac{[S][6][7]}{2}$. Since the 1920s, the thermogravimetric determination of zirconium, as ZrP, has been one of the most widely used quantitative analysis methods [8].

Although the cation exchange properties of ZrP began to be studied in the 1950s, the crystalline zirconium bis(monohydrogen phosphate) monohydrate, α -Zr(HPO₄)₂·H₂O (α -ZrP), was not synthesized until 1964 ^[9]. In 1968, the description of its laminar structure was reported ^[10], for which more precise data appeared in 1969 ^[11]. The structural refinement of α -ZrP was published in 1977 in two independent articles ^{[12][13]}, making it possible to explain the behavior of the material as cation exchangers after determining the position of the hydrogen atoms in the crystalline phase. The α -ZrP consists of a single plane of metal atoms linked through monohydrogen phosphate groups, located alternatively above and below each plane. Each zirconium atom is coordinated with six oxygen atoms of six different phosphates, and each phosphate tetrahedron shares three of its four oxygen atoms with three zirconium atoms, while the fourth is linked to a hydrogen atom, constituting the most active site of the material. The packing of the layers generates six-sided cavities that contain the water molecules, which interact through hydrogen bonds with the P–OH groups in the same plane (**Figure 1**).

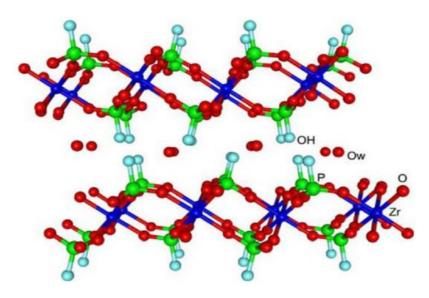


Figure 1. The crystalline structure of α -ZrP. Reprinted with permission from reference $\frac{[14]}{\alpha}$.

1.1. Reported Methods for α -ZrP Synthesis

Three main routes for the synthesis of crystalline α -ZrP have been described: (i) the reflux methods; (ii) the metal complexing reagent-utilizing methods; and (iii) the hydrothermal methods.

The reflux method was first published by Clearfield and Stynes in 1964 $^{[Q]}$. In this method, an amorphous solid is obtained from the direct reaction between phosphoric acid (H₃PO₄) and zirconyl chloride, ([Zr₄(OH)₈(H₂O)₁₆]Cl₈(H₂O)₁₂), that is subsequently washed with distilled water to eliminate all chloride ions, before being refluxed in concentrated phosphoric acid until obtaining crystalline α -ZrP. The degree of crystallinity varies directly with the refluxing duration and the concentration of the used phosphoric acid solutions. Worth noting that almost four decades later, it was demonstrated in researchers' laboratories that the formation of amorphous zirconium phosphate is actually a consequence of washing the precipitate with distilled water $^{[15]}$. This led to the conclusion that crystalline α -ZrP can be directly obtained after the reaction between acid phosphoric and zirconyl chloride at room temperature, although the obtained α -ZrP microcrystals easily hydrolyze in water $^{[15]}$.

On the other hand, the use of complexing agents for the synthesis of highly crystalline α -ZrP was first proposed in 1968 by Alberti and Torracca [16]. This approach, dubbed the HF method, includes the complexation of the Zr(IV) cation with fluoride anions, to form with ZrF₆²⁻ species, and its slow subsequent decomposition in the presence of phosphoric acid. The degree of α -ZrP crystallinity is controlled through the rate of HF elimination. García-Rosales et al. revisited this approach to study the precipitation of α -ZrP from solutions of fluoro complexes of zirconium (IV) using Mexican zirconium sand as raw material [17]. The oxalate anion (C₂O₄²⁻) was also reported to be another complexing agent for zirconium (IV). Horsley et al. [18] and Capitani et al. [19] explored the synthesis of α -ZrP using oxalic acid (HO₂C-CO₂H). Their studies showed that the obtained material had similar structural characteristics and properties to those prepared in the presence of hydrofluoric acid (HF), with the additional advantage of providing more efficient reaction yields in shorter times.

As a third alternative, Shuai et al. proposed the use of a hydrothermal method based on the slow addition (under stirring) of a zirconyl chloride ($[Zr_4(OH)_8(H_2O)_{16}]Cl_8(H_2O)_{12}$) solution to phosphoric acid (H_3PO_4) solutions, followed by subsequent heating of the reaction mixture to 200 °C $^{[20]}$. They observed that the length and thickness of the crystalline α -ZrP $[Zr(HPO_4)_2\cdot H_2O]$ plates are directly proportional to the concentration of phosphoric acid, whereas their length/thickness varies inversely. They also studied the growth rate of α -ZrP crystals as a function of reaction time and phosphoric acid concentration. They found that the process starts with a rapid homogeneous nucleation in the presence of an amorphous phase, in a regime similar to liquid–liquid phase separation that was first reported by Vekilov et al. as the very first step of nucleation $^{[21]}$. This is followed by slow progress, especially at low concentrations of phosphoric acid. This slowness in the metastable stage allows for autoselection of better nuclei and the formation of secondary clusters, from which crystals grow with an observable homogeneity and regularity in size and morphology of the crystals, after dissolving poorly ordered crystals and their reprecipitation (**Figure 2**) $^{[22]}$. This nucleation and crystallization mechanism resembles what was known later in the literature as the two-step nucleation mechanism that was mainly reported for protein crystallization, and very recently, for metal–organic frameworks $^{[21][23][24]}$.

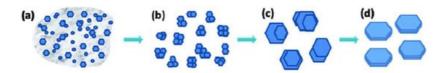


Figure 2. Representation of α -ZrP crystal formation. (a) Precipitation of nanocrystals and amorphous particles from solution. (b) Lattice orientation of nanocrystals with formation of secondary clusters. (c) Growth of microcrystals from secondary clusters. (d) Formation of homogeneous and regular crystals by dissolving poorly ordered areas and subsequent re-precipitation. Reprinted with permission from reference [20], License number: 1173274-1.

Later, Cheng et al. described the synthesis of α -ZrP, but rather in the form of rods, in addition to the usual morphology of plates, as a function of the presence or absence of small amounts of fluoride ions [25]. They showed that in the absence of fluoride ions, mixing zirconyl chloride and concentrated phosphoric acid solutions, that is followed by heating the mixture at 100–120 °C, resulted in the formation of plate-shaped α -ZrP, with a notable increase in crystallinity with increasing temperature. On the other hand, the presence of small amounts of fluoride ions led to the formation of rod-shaped α -ZrP together with the plate-shaped α -ZrP crystals. They attributed the growth of crystals along the crystallographic α -axis to be due to the selective adsorption of fluoride ions on the surface of the plates.

Other unconventional strategies for the synthesis of α -ZrP were reported to have resulted in the formation of microcrystals with different morphologies and textural properties. As an example, Benhamza et al. used a sol–gel synthesis in which a concentrated phosphoric acid solution was added to zirconium (IV) propoxide [Zr(OCH₂CH₂CH₃)₄] in 1-propanol (CH₃CH₂CH₂OH) ^[26]. The aging in the mother liquor increased the degree of crystallinity of the resultant α -ZrP, which varied in structure from amorphous to semi-crystalline materials. Another approach was reported by Hajipour et al. who prepared hexagonal shaped α -ZrP nanoparticles using polyvinyl alcohol [-CH₂CHOH-]_n (PVA) or polyvinylpyrrolidone [(C₆H₉NO)_n] (PVP) in an aqueous solution ^[27]. PVA and PVP acted as complexing organic matrices for Zr(IV) that facilitated the dispersion of α -ZrP particles and controlled their sizes. Afterwards, their (PVA/PVP) elimination through the calcination led to the growth of hexagonally-shaped plates with an average length of 60 nm.

On the other hand, mesoporous zirconium phosphate that bears relatively high specific surface areas and proton conduction capacity $\frac{[28][29]}{[28][29]}$, are usually synthesized by calcination of the reaction product of soluble phosphate and/or phosphonate salts with zirconium oxychloride (ZrOCl₂) or zirconium alkoxide $[Zr(OR)_4]$ $\frac{[30][31][32][33]}{[30][31][32][33]}$. Moreover, Zhao et al. used the reaction between phosphoric acid and zirconium propoxide, in the presence of non-ionic surfactants, to obtain gels that transformed to ZrP with a vermicular morphology after being thermally activated $\frac{[34]}{[35]}$. Furthermore, Tarafdar et al. synthesized spherical zirconium phosphate particles (micro- and meso-porous) with zirconium carbonate $[Zr(CO_3)_2]$ as the source of the metal $\frac{[35]}{[35]}$. In addition to the above, Zhu et al. obtained α -ZrP with unusual textural properties, using an approach consisting of a sol–gel process followed by supercritical drying $\frac{[36]}{[35]}$. For this process, two polymers: polyethylene glycol $[H-(O-CH_2-CH_2)_n-OH]$ (PEO) and polyacrylamide $[(-CH_2CHCONH_2-)]$ (PAM), were used that led to the formation of solids with macropores of controllable size $(0.5-5 \mu m)$. These solids can be modified through altering the relative proportion of both polymers. These materials in monolithic form maintain high porosity, with specific surface areas close to $600 m^2 \cdot g^{-1}$, and good mechanical properties in uniaxial compression.

1.2. Functionally-Adapted α-ZrP Synthesis Routes

The increasing interest in nanotechnology demands the development of new synthetic approaches for laminar materials as α -ZrP, in order to prepare nanoparticles of nanometric dimensions, both in their planes (length/width) as well as in their thickness, until reaching colloidal dispersions of quasi-individual sheets.

In that respective, Bellezza et al. proposed the synthesis of α -ZrP nanoparticles from water-in-oil microemulsions containing zirconyl chloride and phosphoric acid $^{[37]}$. In addition, Alberti et al. synthesized α -ZrP of low crystallinity in a one-step reaction, where a solution of an organic zirconyl salt such as zirconyl propionate $[Zr(CH_3CH_2COO)_4]$, in a polar aprotic solvent (e.g., N,N-dimethylformamide $[(CH_3)_2NC(O)H]$, DMF) is mixed at room temperature with a solution of phosphoric acid in the same solvent, inducing the formation of a transparent dispersion from which the solvent is removed by heating to dryness $^{[38]}$. Although this procedure is chemically effective, the use of organic solvents deprives it from being an ecologically friendly alternative and, thus, biocompatible for medical applications $^{[39]}$. In order to develop synthetic procedures with less environmental drawbacks by avoiding the usage of toxic solvents, ethanol (C_2H_5OH) was used to dissolve zirconyl propionate $^{[40]}$. This led to the formation of gel-like materials containing crystalline domains of α -ZrP sheets intercalated with ethanol. Heat treatment of these gels generates irregularly shaped α -ZrP nanoplates. As expected, the presence of low-coordination phosphate groups $(H_2PO_4^-$ and $H_3PO_4)$ was detected in α -ZrP nanoparticles in a much higher proportion than in massive materials, which favors their reactivity compared to microcrystalline materials.

1.3. Technological Applications of α -ZrP and Their Introduction in Medicinal Chemistry

In addition to its well-known applications as ion exchangers that can assist in purification $\frac{[41][42]}{[42]}$, and proton conductors $\frac{[43]}{[44]}$, or catalysts and catalyst supports $\frac{[14]}{[42]}$, α -ZrP and its derivatives have been explored to obtain different functionalities, such as anticorrosion coating and reinforced mechanical properties.

Owing to its barrier property, exfoliated α -ZrP, after being functionalized with polypyrrole, could be used for the fabrication of waterborne epoxy coatings (PPy-ZrP/WEC), that bear outstanding anticorrosion resistance [45]. In another context, despite the difficulty of introducing inorganic nanoparticles homogenously in a polymeric matrix, different amounts of bromine initiator could be introduced on the surface of α -ZrP nanoplatelets using a dual-epoxide-modified method [46] that facilitated their usage to graft polymethylmethacrylate (PMMA) polymer. This resulted in enforced mechanical properties for the resultant matrix (ZrP-g-PMMA) that could present a new alternative to this PMMA polymer, widely used for biomedical applications [47].

 α -ZrP and its derivatives have also been extensively investigated in biomedical applications. The implementation of nanotechnology for drug delivery processes has become a very promising area of research, with a primary objective to realize the necessity to completely eliminate (or even inhibit) the inherent side effects for the compulsory administration of medications. In this context, lamellar zirconium phosphate can serve as a good choice for being a carrier material in drug delivery systems owing to its excellent biocompatibility [48][49][50][51]. This has allowed for its commercial use in peritoneal dialysis as a sequestrant of ammonium cations [52].

The intercalation capacity of α -ZrP facilitates the incorporation of functional biomolecules into its structure, avoiding their unnecessary interaction with the biological environment, preventing their protein denaturation, and prolonging their lifetime and thus, their biofunctionality. The acidic character of the interlaminar hydrogen phosphate groups favors the reversible intercalation of basic species and, therefore, the ability to sequester and release molecules of biological interest. In general, the expulsion of the molecules contained in the lamellar inorganic matrices is achieved by alterations in the chemical environment, such as a concentration gradient or changes in pH, and other biological stimuli $\frac{|48||49||50||51|}{|48||49||50||51|}$.

Ding et al. were the first to intercalate proteins in microcrystalline α -ZrP [53]. Afterwards, Kumar and McLendon described the intercalation of heme proteins with redox activity [54]. Thereafter, Kim et al. obtained multilayer thin films comprising previously exfoliated α -ZrP sheets and polycations, such as cytochrome c (a monomeric protein that acts as a mitochondrial electronic transporter) [55]. In addition, α -ZrP was also used to immobilize myoglobin, lysozyme, hemoglobin, chymotrypsin, and glucose oxidase [56]. Furthermore, Bellezza et al. immobilized lipase on pristine and functionalized α -ZrP, observing that the amount of lipase retained increases as the hydrophobicity of the surface increases [57]. Moreover, it was observed that the presence of DNA or urea stabilizes the activity of enzymes and proteins in α -ZrP based nanomaterials. For example, the activity of hemoglobin inserted in α -ZrP is greater when it is co-intercalated with DNA [58], while moderate concentrations of urea stabilize heme proteins (methemoglobin and metmyoglobin) retained in α -ZrP, thus increasing its activity as well as its average lifetime by alterations in the chemical environment [59].

A quite different biologically relevant application is the one that has been reported by Xu et al., who selectively captured phosphopeptides in complex mixtures using α-ZrP nanoplates, in order to analyze tryptic digestions of mouse liver and leukemia cell phosphoproteomics, identifying 158 phosphopeptides (209 phosphorylation sites) of 101 phosphoproteins in mouse liver lysate, and 78 phosphopeptides (104 phosphorylation sites) of 59 phosphoproteins in leukemia cell extract $[\underline{60}]$. On the other hand, Díaz et al. studied the controlled release of nanoencapsulated insulin in α -ZrP, using pH variations as a stimulus [48]. Afterwards, they reported the use of zirconium phosphate nanoplates for the encapsulation of cisplatin (a well-known drug in chemotherapy treatments against various types of cancer) and its administration to tumor cells [51]. Cisplatin, which was inserted using ion exchange, was tested in vitro for cytotoxicity against human breast cancer (MCF-7) with promising results. The intercalated drug was released only at low pHs, typical for tumor cells, which should minimize the side effects of cisplatin on noncancerous tissue. In addition, these researchers used α-ZrP to intercalate doxorubicin (DOX), a cytostatic from the anthracycline family, widely used in cancer chemotherapy, demonstrating its effectiveness against MCF-7 cells compared to free DOX, in a process that leads to the uptake of nanoparticles by endocytosis, reaching and penetrating cancer cells to a greater extent than healthy cells [50]. In this mechanism, the hydrogen-mediated phospholipid cell membrane phosphates participate with P-OH groups on the surface of α-ZrP. Studies by others on DOX@α-ZrP demonstrated that the zirconium phosphate nanoparticles are highly hemocompatible and do not show hemolytic activity towards human red blood cells [49]. González et al. conducted in vitro studies showing that α -ZrP nanoplates deliver DOX to cancer cells, while DOX retains its original anticancer activity [61]. Furthermore, while DOX induces the generation of oxygen radicals in both cancer cells and healthy mammary epithelial cells, DOX@a-ZrP causes greater oxidative stress in MCF-7 breast cancer cells than in MCF-10A breast epithelial cells, thus showing

selective toxicity towards cancer cells. Subsequent studies have reported the success of modifying the surface of the zirconium phosphate in DOX@ α -ZrP, improving the ability to administer DOX, and increasing its biocompatibility [62].

Another variety of α -ZrP (3-aminopropyl)triethoxysilane [H₂N(CH₂)₃Si(OCH₂CH₃)₃] (APTES) functionalized zirconium phosphate nanoparticles has also been used for drug delivery. Li et al. used APTES@ α -ZrP to anchor hyaluronic acid (HA) to combine the carrying capacity of the drug in zirconium phosphate and the tumor targeting capacity of HA in the administration of paclitaxel (PTX), a drug used in the treatment for cancers of the lung, ovary, breast, and advanced forms of Kaposi's sarcoma [63]. In vivo experiments in A549 lung cancer showed good antineoplastic ability and minimization of the toxic side effects of PTX.

Donnadio et al. prepared materials with antimicrobial and antibiofilm activity using nanocrystalline α -ZrP as a carrier for chlorhexidine (CLX), which was used to prepare carboxymethylcellulose (CMC)-based composite polymeric films as wound dressings with reduced cytotoxicity, in order to release the antiseptics in a prolonged way, showing good antibacterial activity against Gram-positive and Gram-negative bacteria [64]. CMC@ α -ZrP films are uncytotoxic to human dermal fibroblast and keratinocyte cell lines. When CLX was incorporated into CMC@ α -ZrP, its cytotoxicity was significantly reduced when compared to free CLX, probably because of the controlled release of CLX that kept its concentration at low levels.

Very recently, Adawy et al. implemented the hydrothermal methodology to synthesize monohydrated nanolaminar α -ZrP and subsequently enriched their nanolayers with silver nanoparticles (AgNPs) of uniform dimensions and distribution [65]. The structural stability of α -ZrP was confirmed before and after enrichment with AgNPs as well as the assessment of their antimicrobial activity and the cytocompatibility. They reported that the pristine α -ZrP did not show any cytotoxic effects against Sarcoma osteogenic cells, with a small loss of its biocompatibility when enriched with silver, yet with outstanding antimicrobial effectiveness against *Escherichia coli*.

2. Titanium Phosphates

Although zirconium phosphates have so far been the most studied lamellar solids of all tetravalent metal phosphates since 1950s, titanium laminar phosphates have also attracted particular attention around a decade later (1960s). Alberti et al. [16][66], following the wake of the seminal investigations on zirconium phosphates by Clearfield et al. [9], were pioneers in the study of crystalline titanium phosphates. In addition, the contributions of the Soviet school are also noteworthy in this field, with special mention to Chernorukov et al. [67]. In this scientific environment, researchers' working group at the University of Oviedo (Research Group of Synthesis, Structure and Technological Application of Materials [SYSTAM], https://systam.grupos.uniovi.es/) published its first articles on titanium phosphates in the early 1980s [68][69][70]. The established synthetic routes for the layered titanium phosphate $[\alpha-Ti (HPO_4)_2 \cdot H_2O]$ are very similar to those of layered zirconium phosphate [α-Zr(HPO₄)₂·H₂O]. Only recently new methodologies have been introduced, mostly based on the hydrothermal synthesis, but rather using other compounds as precursors, such as protonated layered lepidocrocite-type $K_{0.7}(Ti)_2O_4$ [71]. Unlike zirconium phosphates, titanium phosphates can be also synthesized in another morphology, exhibiting the form of nanorods, and thus are known as nanofibrous titanium phosphates. The most popular synthetic route of nanofibrous titanium phosphates is also the hydrothermal method at temperatures 160-250 °C, using a titanium precursor such as titanium tetrachloride ($TiCl_4$) in aqueous phosphoric acid solutions [72]. Moreover, several studies have reported on the possibility for growing a nanofibrous titanium phosphate thin film on titanium substrates after hydrothermally treating the latter in phosphoric acid solutions at 180 °C [73] or 250 °C [74], but also at 120 °C in the presence of hydrogen peroxide as a catalyst [75]. Herein, researchers provide further details on the different titanium phosphate phases, categorizing them into nanolayered and nanofibrous phases.

2.1. Nanolayered Titanium Phosphates

So far, there are three nanolayered titanium phosphates reported in the literature. The crystalline titanium bis(monohydrogen phosphate) monohydrate, α -Ti (HPO₄)₂·H₂O (α -TiP), is an iso-structure of its zirconium counterpart (α -ZrP) [76]. y-titanium phosphate (y-TiP), formulated as y-Ti(HPO₄)₂·2H₂O and later as y-Ti(PO₄)(H₂PO₄)·2H₂O, is another interesting dihydrate layered material that is considered as an analogue to α -TiP, with larger interlayer spacing than α -TiP since it comprises two interlayer water molecules. y titanium phosphate (y-TiP) is also an iso-structure of y zirconium phosphate, so do its dehydrated form known as β -TiP [β -Ti(PO₄)(H₂PO₄)] and β -ZrP [β -Zr(PO₄)(H₂PO₄)] [77][78][79]. As expected, α -TiP and y-TiP have many profound applications, such as ion exchange [80], catalysis [30], ionic conductivity [44] [81], and electrodes for Li- and Na ion storage [82][83][84], in addition to their capability of being functionalized with organic species and also being exfoliated [85][86][87]. Nevertheless, there is almost no record for their biologically relevant applications.

2.2. Nanofibrous Titanium Phosphates

Several titanium phosphate compounds have been reported with different morphologies and thus different plausible applications $\frac{[88][89][90][91]}{}$. With a nanofibrous morphology, two polymorphs with the chemical formula: $Ti_2O(PO_4)_2 \cdot 2H_2O$, called π -TiP and ρ -TiP, were synthesized around two decades ago [72]. Although π -TiP and ρ -TiP are compounds with neutral lattices, they exhibit an outstanding ability to incorporate intracrystalline metal cations through ion exchange processes [92][93]. Particularly, π-TiP reaction with alkali metal nitrates in the molten state results in the formation of fibrous phases charged with the reactant alkali metal $\frac{92}{2}$. In terms of their crystallographic structures, the structure of ρ -TiP was reported shortly after being solved ab initio from synchrotron X-ray and neutron powder diffraction data (Figure 3a) [94]. On the other hand, there were only some scientific hypotheses reported about the most probable structural ordering of π -TiP $\frac{[92]}{}$. The crystal structure of the ρ -TiP phase has two channels that run parallel to the crystallographic a-axis in which the water molecules are housed. Recently, researchers succeeded to reveal the crystal structure of π -TiP [95]. The π -TiP phase, obtained under mild hydrothermal conditions, crystallizes in the monoclinic system (space group P2₁/c) in contrast to the previously described ρ -TiP (triclinic, P-1). The π -TiP structure is made up of TiO₆ and TiO₄(H₂O)₂ octahedra, connected with orthophosphate groups, defining the anisotropic three-dimensional packing, consistent with the existence of preferential crystal growth directions. The properties of both polymorphs have scarcely been explored, probably because they were just synthesized less than 3 decades ago. Although both π - and ρ -TiP exhibit low accessible porosity when hydrated (11-16 m²·g⁻¹) [96], Amghouz et al. showed that this can be changed significantly if they are treated thermally. In such a process, the crystallized water molecules are eliminated, and the resultant dehydrated phases exhibit an unusual adsorption capacity for nitrogen a little above ambient temperature [97]. For the ρ-TiP, the thermal treatment at temperatures of 200-300 °C results in the generation of tetrahedral (ex-octahedral) titanium atoms (Figure 3b) that are responsible for the enhanced nitrogen adsorption capability $\frac{[97]}{}$. On the other hand, this treatment for the π -TiP was shown to lead to a change in the coordination environment of one of the titanium atoms, which goes from octahedral to tetrahedral, with the formation of the anhydrous $Ti_2O(PO_4)_2$ [97]. Beside their nitrogen adsorption properties, π -TiP has been recently investigated for its proton conductivity. π -TiP was doped in a chitosan matrix to study the proton conductivity of the resultant chitosan-based composite membranes (CS@ π -TiP) [95]. This study showed that doping the CS matrix with π -TiP (5 w/w%) results in a 1.8-fold rise in the proton conductivity compared to the bare membrane [95]. Another explored property is their attraction to water. Yada et al. prepared super-hydrophilic thin films made up of micro- and nano-crystals of π -TiP with controlled morphologies that transform to superhydrophobic layers when π -TiP is modified with alkylamines [75]. Similarly, Cai et al. showed that TiP thin films grown on titanium substrates can possess a hydrophilicity that can be switched to a super-hydrophobicity through increasing the preparation temperature [98]. Recently, π-TiP has been reported to possess ultrastability as a Ca²⁺ storage material with a minimal dimensional change and almost no transformation in the crystallographic structure upon the insertion or extraction of Ca²⁺ [99]. This exceptional stability that stands for over 1700 cycles of the Ca^{2+} insertion and extraction presents π -TiP as a viable electrode material for calcium-ion battery applications [99]. Therefore, researchers can estimate that many attempts could be made in the near future to utilize these nanofibrous phases for some of the other applications that have been established for the nanolayered titanium phosphates.

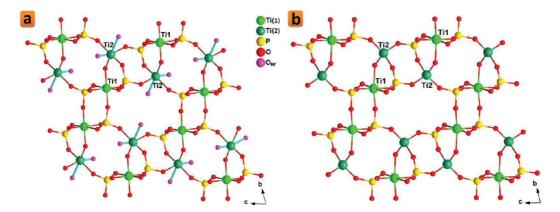


Figure 3. Crystalline structure of the dihydrate (a) and anhydrous (b) phases of ρ-TiP. Reprinted with permission from reference $\frac{[86]}{}$, License number: 1173277-1.

2.3. Prospective Applications of TiP in the Biomaterials Research Field

Since α -TiP is an iso-structure of α -ZrP, and has been reported to share many of its physical and chemical properties beside its non-spherical layered morphology, it should have potential applications for being used in biological applications $\frac{[100]}{100}$. Similar to α -ZrP, one of the most important aspects to consider for this inorganic laminar nanomaterial in biological applications lies in its ability to provide a controlled release of drugs or charged nanoparticles, while its non-spherical

morphology favors its adherence and margination to biological tissue $\frac{[101][102]}{[101]}$. Therefore, just like α-ZrP, α-TiP could be an interesting alternative for drug delivery and bone cement applications, especially if it is enriched with biofunctional nanoparticles that could improve its antimicrobial properties $\frac{[103][104]}{[103]}$. In this context, Adawy et al. recently used the hydrothermal methodology to synthesize monohydrated nanolayered α-TiP, whose surfaces were, afterwards, enriched with AgNPs of uniform dimensions and distribution $\frac{[65]}{[65]}$. After the structural evaluation of the resultant Ag-enriched α-TiP phase, assessments of the cyto-viability and antimicrobial efficacy were performed. Surprisingly, it was reported that α-TiP bears an outstanding cytocompatibility that even surpasses that of α-ZrP, particularly after being enriched with AgNPs $\frac{[65]}{[65]}$. More interestingly, α-TiP appeared to possess higher intercalation capacity for silver ions compared to that of α-ZrP, while, as predicted, AgNPs-enriched α-TiP possessed a perfect antimicrobial activity $\frac{[65]}{[65]}$. Therefore, these recent results highlight the potential capabilities of α-TiP, especially under stressful biological conditions, in which the decrease in pH values can stimulate more release of the functional silver ions that are richly intercalated in the interlayers of α-TiP.

On the other hand, nanostructured inorganic compounds, such as nanofibrous ρ -TiP and π -TiP, have generated increasing interest in the research directed towards the synthesis of biomaterials. These compounds are usually prepared in advance and used as bone cement or applied as a surface layer covering the implant [105][106][107][108][109], although it is also possible to induce its growth on the substrate surface [110][111][112]. In the latter case, the thin layer that constitutes the coating acts as a messenger through which interaction with the surrounding biological system occurs [113][114][115]. The outer layer of the device should be compromised of compounds naturally present in the biological environment, commonly calcium phosphate phases [116][117][118][119]. However, nanostructured systems are also used for this application using the metal of the (metallic) substrate as the source of the inorganic matter [120][121]. A good example of these metals is titanium, with high osseointegration capacity and ease of physical attachment to bone, owing to the spontaneously formed oxide layer on its surface, which does not cause denaturation of proteins in the proximity of the implant $\frac{[122]}{}$. Thus, the formation of coating layers of nanostructured compounds containing titanium on the metallic surface of titanium has been described with positive results [123][124], including its use, together with its widely used alloy, titanium-aluminum-vanadium alloy (Ti-6Al-4V), in implant applications [125][126]. Modifications of the surfaces of titanium and its alloys by coating them with a layer of nanostructured materials could promote their hydrophilicity, surface roughness, biocompatibility, and bioactivity, by improving both their functionality and their osseointegration and, therefore, their long-term fixation of the implant. In this respect, the two polymorphs π - and ρ -TiP are nanostructured compounds that obviously contain titanium, the former of which could be grown successfully on titanium surfaces [73][74][75].

Recently, Adawy et al. synthesized nanorods of the polymorphs π -TiP and ρ -TiP and enriched them with AgNPs that provided outstanding antimicrobial activity without much of a compromise to their cytocompatibility ^[127]. Additionally, they hydrothermally treated the commercially available Ti–6Al–4V alloy in aqueous phosphoric acid solution in attempts that led to the growth of a nanofibrous coating layer of π -TiP on the surface of Ti–6Al–4V alloy discs, where the size as well as the surface roughness of this nanofibrous π -TiP coating layer can be controlled ^[127]. Although intercalation is not an option for the nanofibrous morphology of π -TiP, Adawy et al. reported the enrichment of the coated surfaces with AgNPs and their subsequent doping with strontium ions, which in turn played an outstanding role in controlling the release of the antimicrobial AgNPs, in a mechanism that improved the cytocompatibility and long-term biofunctionality of the resultant coated biomedical alloy as they described in their long-term ion release assessments ^[127].

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