

Halide Perovskites

Subjects: [Nanoscience & Nanotechnology](#) | [Engineering, Electrical & Electronic](#)

Contributor: Soon-Gil Yoon

Halide perovskites (HPs), with an excellent photoactive nature, dielectric, piezoelectric, ferroelectric, and pyroelectric properties, have been potential candidates for obtaining flexible nanogenerator-based self-powered sensors including light, pressure, and temperature. Additionally, the photo-stimulated dielectric, piezoelectric, and triboelectric properties of HPs make them efficient entrants for developing bimodal and multimode sensors to sense multi-physical signals individually or simultaneously.

halide perovskite

nanogenerator

PENG

PyENG

TENG

self-powered sensor

temperature sensor

pressure sensor

physiological sensor

photodetector

1. Introduction

Halide perovskite materials consist of three-dimensional ABX_3 -type crystal structures (**Figure 1**, crystal structure). In ABX_3 , A denotes the monovalent organic or inorganic cation (such as $MA^+ = CH_3NH_3^+$, $FA = CH(NH_2)_2^+$ and Cs^+), B represents the divalent cation (such as Pb^{2+} , Sn^{2+} , and Ge^{2+}), and X is the halide anion (Cl, Br, I). In the crystal structure, cation A is coordinated with 12 neighboring X, and cation B is connected with 6 X anions, forming cuboctahedral and octahedral geometries, respectively. The structural formability of the 3D- ABX_3 HPs and their stability were determined by the Goldschmidt tolerance factor (t) and the octahedral factors (μ) [1][2]. The tolerance factor, $t = (r_A + r_X)/\sqrt{2}(r_B + r_X)$, where r_A , r_B , and r_X represent the effective ionic radii of A, B, and X in the HP, respectively. The r_B/r_X is defined as an octahedral factor (μ) and is directly correlated with a BX_6 octahedron, playing an important role in the multifunctional properties of HP materials. The tolerance factor values for HP materials were found to be within the 0.8–1.11 range, while the octahedral factor values were in the range of 0.44–0.90 [3][4]. The calculated t values for cubic, orthorhombic, and hexagonal structured HPs were $0.8 < t < 1.0$, $t < 0.8$, and $t > 1$, respectively [2][5]. Numerous methods have been proposed to fabricate HP thin films, such as the one-step solution process, two-step sequential deposition, vapor deposition method, and vapor-assisted solution process [6][7][8][9]. Solution methods have been promoted due to simple and low-temperature fabrication ways and their compliance with flexible and large-scale devices [10][11].

Furthermore, the solution methods are promising to control crystalline formation better, leading to tunable properties at nano/micro-scale [12][13]. These HP-based materials have been considered to fabricate flexible nanogenerators owing to their impressive dielectric, ferroelectric, piezoelectric, and pyroelectric properties and beneficial features such as low-temperature and large-scale fabrication routes and controllable properties [14][15][16][17][18]. In particular, the dielectric and piezoelectric properties of HP-based materials are remarkable and nearly

similar to those of ceramic materials [11][13][14][19][20][21]. Generally, the dielectric material is electrically an insulator and will readily polarize under an applied electric field. The ability of a material to store electrical energy can be estimated by its dielectric permittivity (ϵ_r) values. However, the dielectric loss factor describes the inefficiency of a material to hold the stored energy. Consequently, a material with a high loss factor cannot hold the charge completely indicating leaky nature of the sample. Therefore, materials with larger dielectric constant values and low dissipation factor values tend to generate higher output performance in nanogenerator applications because the nanogenerators are designed as part of capacitors [20]. Many researchers have demonstrated the dielectric properties of HP-based materials and also systematically examined the changes in dielectric properties of compositionally tuned HP materials [11][13][14][21]. The dielectric polarization mechanism in HP-based materials is nearly similar to well-known ceramic perovskite materials. The existing organic cation at the center of an HP-crystal structure (e.x., MA^+ cation in MAPbI_3) disorders under an applied field; thus, resulting in a dipolar polarization in HPs [11].

As depicted in **Figure 2a**, the most widely used MAPbI_3 perovskite revealed an ϵ_r value of ~ 52 and low dissipation value of ~ 0.02 at an applied frequency of 100 kHz [19]. Moreover, FAPbI_3 perovskite exhibited an improved dielectric property due to the presence of a larger cation (FA^+) that has larger dipole momentum compared to that of MA^+ cation [22]. In addition, the dielectric properties of HPs can be significantly enhanced by partial tuning of its composition. For example, the reported halide-doped MAPbI_3 perovskite samples displayed a notable enhancement in dielectric properties upon halide doping (either Cl or Br) [14]. In particular $\text{MAPbI}_{3-x}\text{Cl}_x$ displayed a high ϵ_r value of ~ 90.9 . The improved property is not only because of improved grain size but also the existence of binary systems such as MAPbI_3 and MAPbCl_3 in the final film which led to interfacial polarization between two phases. Similarly, the Fe-incorporated MAPbI_3 unveiled enhanced ϵ_r value of ~ 107 upon partial incorporation of Fe ($x = 0.07$) [13]. The improved dielectric properties promoted the enhancement in piezoelectric output performance of the $\text{MAPb}_{1-x}\text{Fe}_x\text{I}_3$ based PENG. Furthermore, the lead-free Sn-based HP materials also possess high dielectric properties [23].

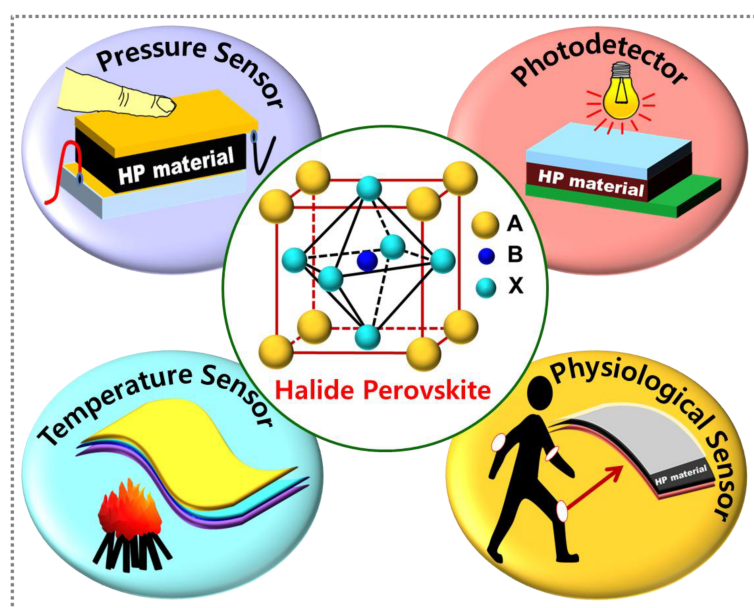


Figure 1. Schematic illustration of ABX_3 -type halide

perovskite structure and their utility in nanogenerator-based self-powered sensing applications.

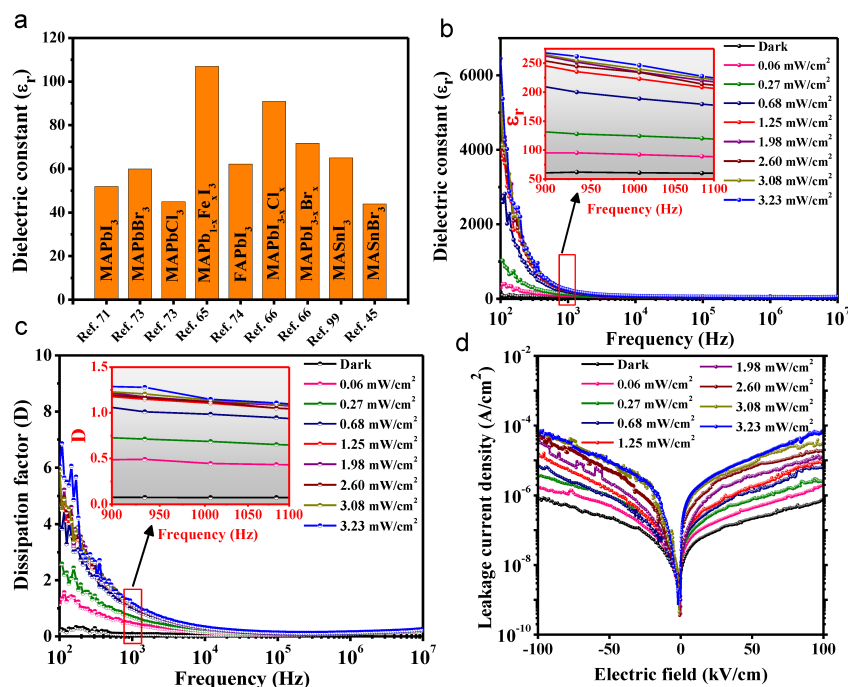


Figure 2. (a) Dielectric constants of various

frequently used HP materials. Light intensity effect on frequency-dependent (b) dielectric constant and (c) dissipation factor of MAPbI₃-PVDF composite films. (d) Light intensity effect on leakage current density of MAPbI₃-PVDF composite films as a function of applied electric field. (b–d) Reprinted with permission from ref. [24], Copyright 2020, American Chemical Society.

2. Halide Perovskite Nanogenerator-Based Self-Powered Bimodal Sensors

Recently, several emerging sensor technologies have been developed to detect single-mode external stimuli, such as light, pressure, temperature, and humidity [25][26][27][28]. However, emerging a flexible single sensor unit capable of sensing multiple signals simultaneously is of great significance. Stacking or incorporating single-mode sensor units to build a multimode sensor can be constrained by the signal interference of different sensor units and disparities in their mechanical properties [29][30]. Therefore, it is of prodigious importance to fabricate a flexible single-structure bi/multifunctional sensor capable of perceiving various external signals using flexible materials in simple and low-cost methods. Hence, HP materials are the potential entrants to develop bi/multifunctional sensor devices in facile ways owing to the combined properties such as optoelectronic, piezo/ferroelectric, pyroelectric, and thermoelectric properties [12][19][31][32][33]. Eom et al. constructed the first HP material-based self-powered bimodal sensor (i.e., pressure cum light sensor) covering the PENG structure of glass/AAA/MAPbI₃/SiO₂/AAA (Figure 3a) (Here, AAA is AZO-Au-AZO multilayer) [34]. The MAPbI₃ perovskite thin film was prepared using the chemical vapor deposition (CVD) technique to study its long-term air-stability. The sputtered thin SiO₂ layer helped to prevent the electrical short-circuit of the device. The nanogenerator poled at 60 kV/cm exhibited a linearly increasing trend in piezoelectric output with increasing applied pressure from 5 to 70 kPa and obtained the pressure sensitivity value of ~8.43 mV/kPa (see Figure 3b). The device was simultaneously exposed to light while applying strain (30 kPa) to explore the bimodal sensing (pressure and light) features of the nanogenerator. It was observed that the device displayed a notable dependency on the intensity of focused light (Figure 3c), signifying

the ability of the device to identify different intensities of light in a self-powered way. Another research group has also verified the multimode sensing (i.e., light and pressure signals sensing) ability of inorganic HP material-based TENG [35]. The reported light-stimulated e-skin-based TENG has the structure of PET/PEDOT: PSS/CsPbBr₃/Au. It was operated based on combined photoelectric and triboelectric effects of CsPbBr₃ material, as illustrated in **Figure 3d**. Under repeated vertical contact-separations of the e-skin by finger tapping, the triboelectric output was notably increased with increasing applied force, indicating the low-level pressure sensing ability of the nanogenerator in a self-powered way (**Figure 3e**).

Additionally, authors have used the same device to identify the various intensities of light by illuminating the e-skin using UV-light (~365 nm) for 8 s at an applied force of 0.3 N. They observed the light-dependent responsivity of the device, as shown in **Figure 3f**. Furthermore, it is believed that the e-skin can be successfully implemented as a memristor to comprehend the human-machine interfaces in emerging artificial intelligence systems. Improving the long-term stability of HPs is imperative and remains a key challenge irrespective of their great potentiality as active sensing materials. Recently our group has found the tremendous bimodal sensing application of HP-polymer composite by emerging long-term stable single-structure device using a simple and cost-effective approach [24]. The flexible single-structure multifunctional device capable of mechanical energy scavenging and simultaneous sensing of multi-physical signals—light and pressure—was constructed using spin-coated MAPbI₃-PVDF composite film onto the flexible plastic substrate followed by sputtering of gold-inter digitated electrodes (gold-IDE) (**Figure 3g**). The reported nanogenerator was successfully operated as an energy harvester. Furthermore, it was simultaneously operated as a self-powered light and pressure sensors in both piezoelectric and triboelectric modes of operation owing to the combined photovoltaic and piezo/triboelectric effects of the active composite film. It was observed that the piezoelectric output of the device under a dark state increased linearly with increasing applied pressure. It displayed high output voltage of ~33.6 V at an applied pressure of 300 kPa, making the device a potential candidate to be used as a self-powered pressure sensor with excellent pressure sensitivity of 0.107 V/kPa (**Figure 3h**).

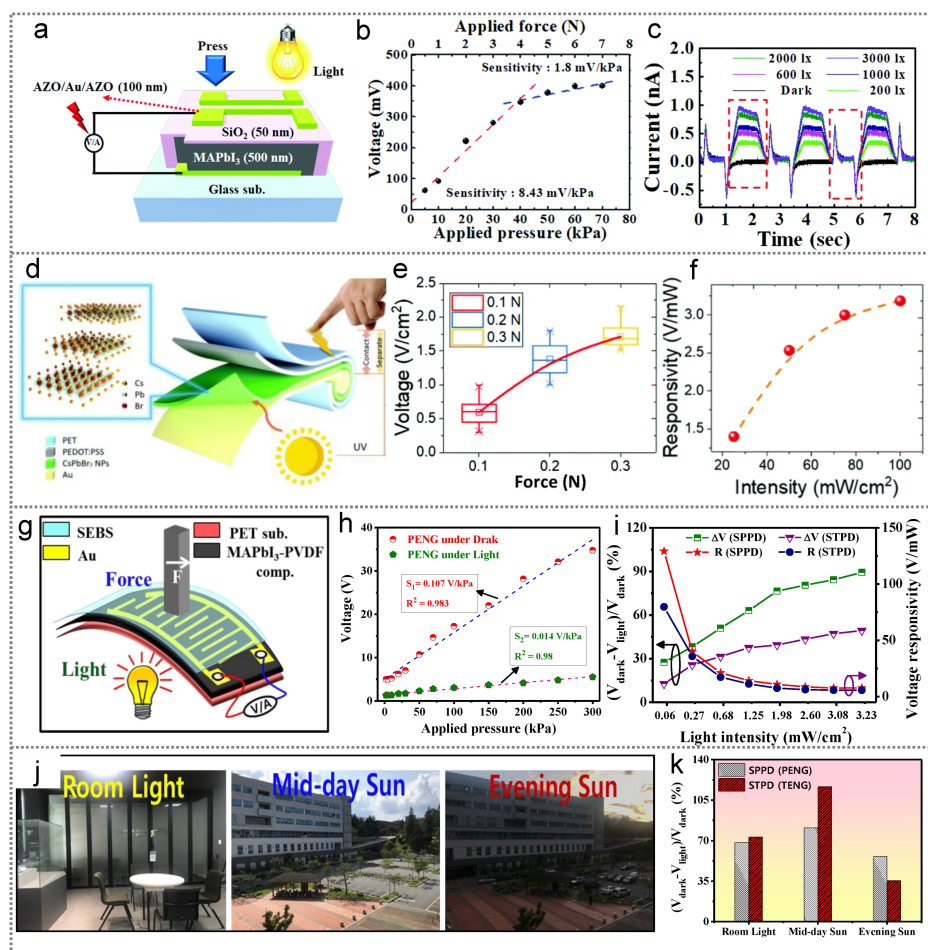


Figure 3. (a) Schematic illustration of

MAPbI₃ PENG-based bimodal sensor, (b) corresponding pressure dependent output voltages revealing the pressure sensitivity, and (c) light-controlled piezoelectric output current signals at an applied constant pressure of 30 kPa. (a–c) Reprinted with permission from ref. [34], Copyright 2018, The Royal Society of Chemistry. (d) Schematic representation of CsPbBr₃ TENG-based bimodal sensor, (e) corresponding pressure dependent output voltage, and (f) responsivity of the device under various light intensities. (d–f) Reprinted with permission from ref. [35], Copyright 2021, The Royal Society of Chemistry. (g) Schematic sketch of MAPbI₃-PVDF-based bimodal sensor in PENG mode, (h) corresponding pressure sensitivity under dark and illumination, (i) corresponding voltage change and responsivity with respect to light intensities, (j) digital images of various light environments used for real-time operation of the device, and (k) corresponding change in voltages. (g–k) Reprinted with permission from ref. [24], Copyright 2020, American Chemical Society.

3. Conclusions, Outlooks, and Opportunities for Future Development

HPs have been recognized as potential entrants in numerous research fields, including solar cells, LEDs, nanogenerators, and photodetectors. Among these, the nanogenerators are the recent application of HPs. They are highly sensitive to various external perturbations, such as pressure, temperature, and light. Hence, controlling the performance of a nanogenerator under external perturbations such as temperature and pressure predominantly enabled it to be used as a self-powered temperature or pressure sensor, respectively, without dependency on a

battery. Additionally, illuminating nanogenerator significantly changes their output performance owing to the combined optoelectronic properties with piezoelectric property, or the triboelectric effect. Such light-stimulated performances of nanogenerators prominently enabled them to be used as self-powered photodetectors to detect the type of light or quantity of light intensity. This progress report briefly discussed the structural, piezoelectric, pyroelectric, and triboelectric properties of nanogenerators of HP materials. After that, we discussed the working mechanisms of several nanogenerators, including PENG, TENG, and PyENG as self-powered pressure and temperature sensors. We discussed the operation of PENG and TENG, as self-powered photodetectors under light illumination and present the possible underlying mechanisms to deliver the light-controlled outputs. Thereafter, we provided an overview of the exploration of using HP-based nanogenerators to sense various external factors such as pressure, temperature changes, and light. It is seen from the examples discussed above that when a nanogenerator is subjected to a controlled exposure to these external stimuli, continuous output voltage and current change linearly, which is a prerequisite for realizing reliable sensing systems. The recently emerging bimodal/multimodal sensing applications of HPs were also reviewed in addition to the single-mode sensor devices. We suggest the future projections of the research towards developing single-structure multimodal sensing devices.

Improving sensitivity, performance, and long-term stability of HPs is challenging despite many studies involving HPs as active sensing materials in self-powered nanogenerator-based sensors. Therefore, we identify potential directions and opportunities for future research. Recently, numerous methods, such as surface passivation, compositional tuning, fine-doping, and polymer composites, have been used to improve the air-stability of HP materials [23][24][16]. Furthermore, it has been demonstrated that the layered 2D-HPs have significantly good air and moisture stability than the 3D-HPs because of long-chain organic molecules [36][37]. Additionally, these 2D-HPs offer extremely high piezo/ferroelectric properties, helping to construct flexible, high-performance nanogenerators and reliable sensors with high sensitivity and stability [38]. Additionally, the development of eco-friendly HP material-based self-powered sensors has been highly prioritized to alleviate the environmental and human health risks. Additionally, extending the investigation to combine three or more properties may be helpful in developing trimodal/multimodal sensors to detect multiple physical signals simultaneously using a single-structure device without any signal interference. For example, our group has reported a single-structure TPS-fusion generator capable of harvesting multiple energy sources, such as thermal, mechanical, and light energies owing to the multiple properties of comprised MAPbI₃ material [39]. Finally, the flexible and stretchable self-powered sensing systems have significant potential to be for e-skin and wearable electronic device applications in the future. In this regard, considerable research should be conducted on HP materials to develop self-powered stretchable multifunctional devices that can continuously operate in harsh environs.

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